

Metastable High Entropy Alloys

Max-Planck-Institut für Eisenforschung



Short version

Max-Planck High Entropy Alloy Team



Rao



Magnetism

Hydrogen Embrittlement

Luo



Corrosion

Kasian

Electro-Catalysis

Phase transformation

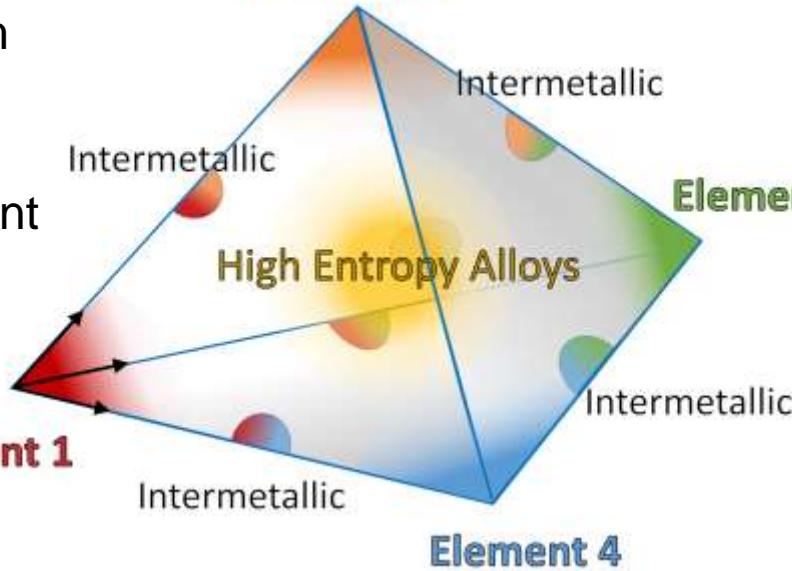
Scheu

Li

Springer

Combinatorial Metallurgy

Element 2



Jägle



Additive manufacturing

Ponge



Lu

Mechanical properties

Dehm

Electron microscopy



Atom probe

Liebscher

Thermodynamics & kinetics

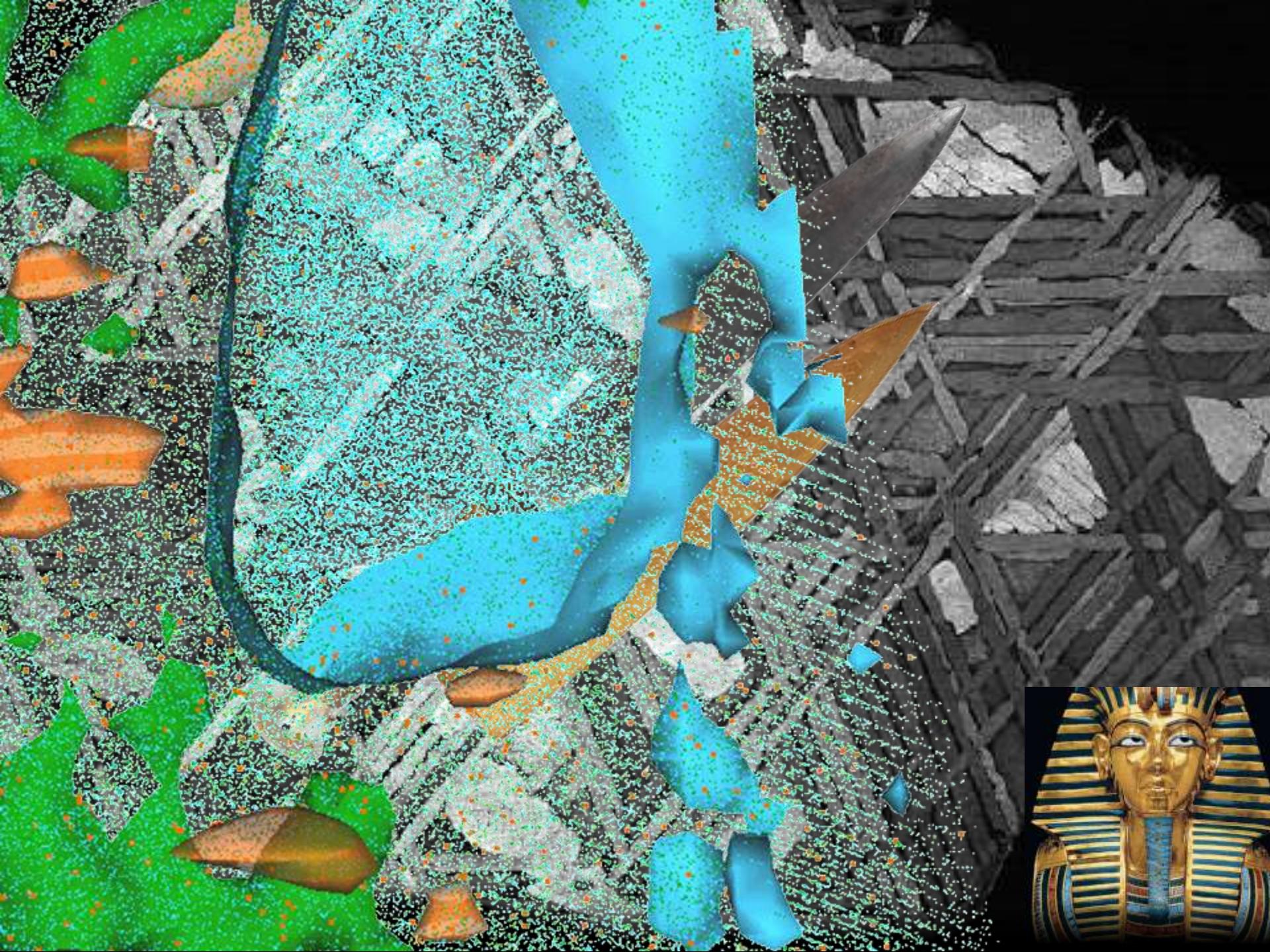


Gault



Grabowski

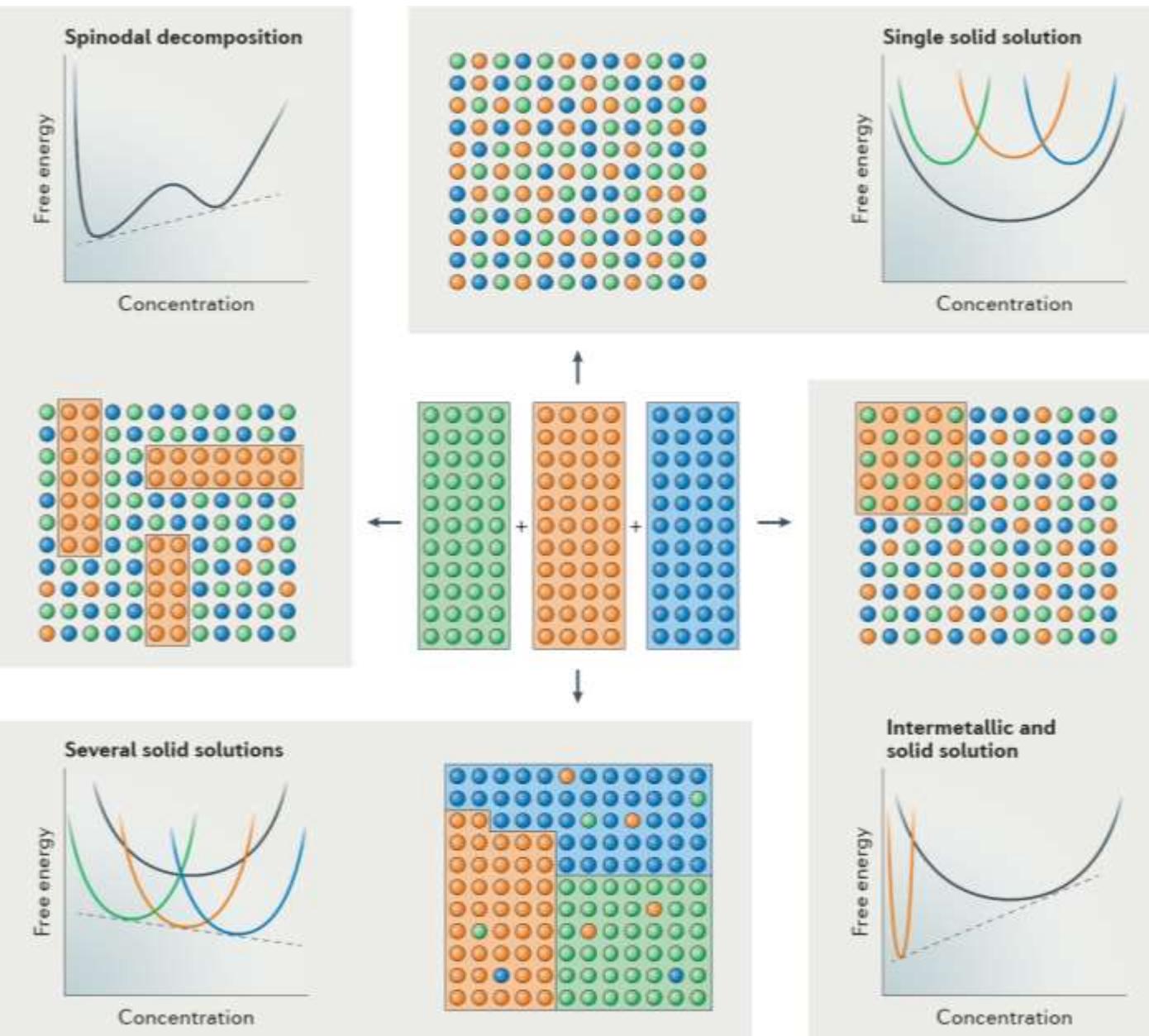
Neugebauer Körmann



- Global metal market: 3000 billion € / year
- Green energy supply
- e⁻ and H in transport and industry
- Sustainable production and CO₂ reduction



Alloy thermodynamics

100
YEARS 1919–2019

Metastability high entropy alloy design



Mechanistic alloy design

Mechanism selection

- Twinning
 - Martensite
 - Precipitates
- ...

Effects

- TWIP
 - TRIP
 - Multiphase
 - Ultra fine grained
 - Dislocation patterning
 - Nano precipitates
 - Phase metastability
 - Metallic glass
 - Interstitials
 - Spinodal
 - Maraging
- ...

Mechanistic high-entropy alloy design

Mechanical Metastability
Chemical Metastability



- TWIP HEA
 - TRIP HEA
 - Spinodal HEA
 - Maraging HEA
 - Metallic glass HEA
 - Interstitial HEA
 - Dual-phase HEA
 - Microalloyed HEA
- ...

High-entropy alloy design

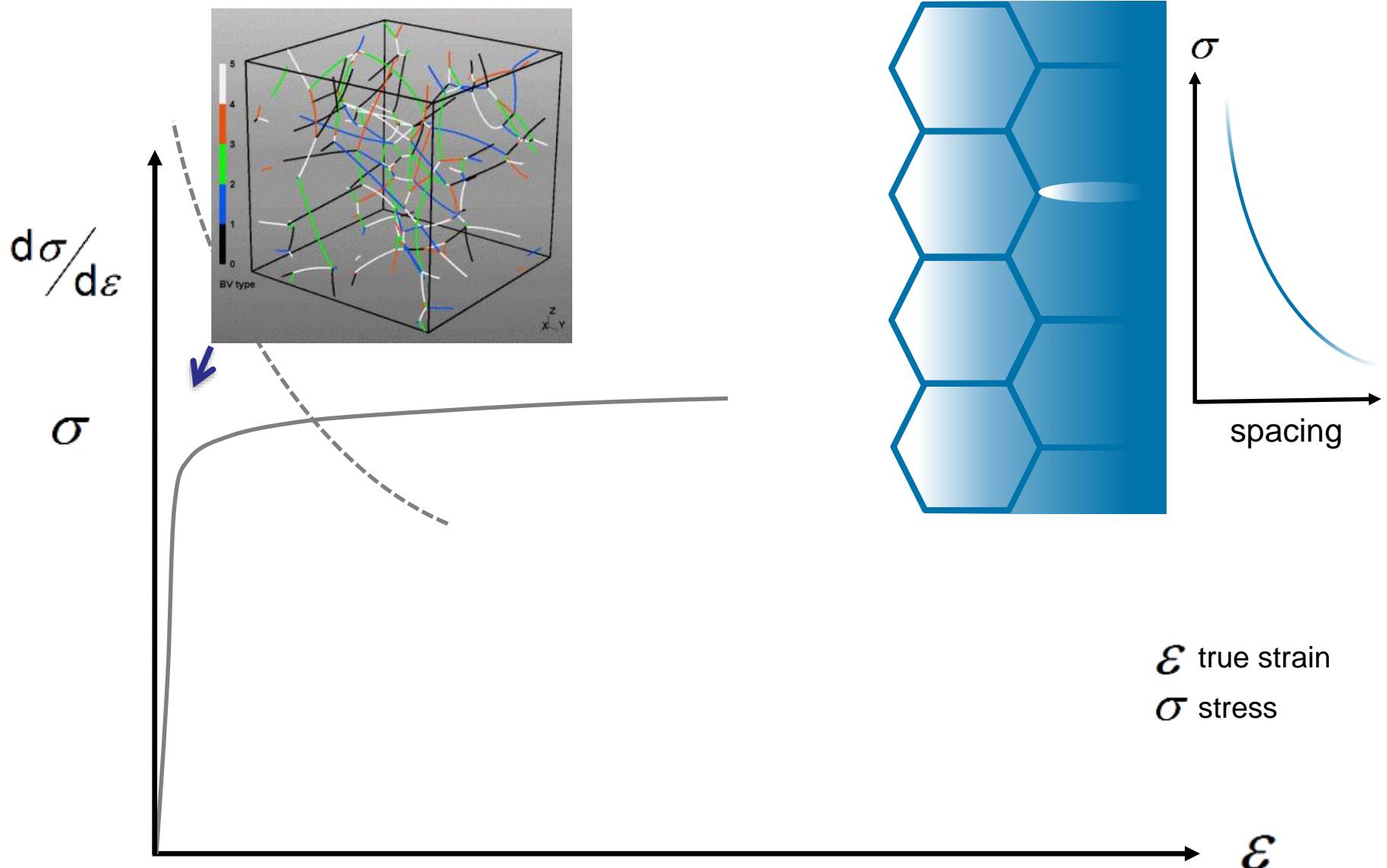
Compositional tuning

- Stacking faults
 - Spinodal
 - Misfit
- ...

Effects

- Cottrell clouds
 - Stability by entropy
 - Massive solid solution
 - Dislocation decoration
 - Gibbs segregation
 - Intermetallics
 - Suzuki effect
 - Precipitation
- ...

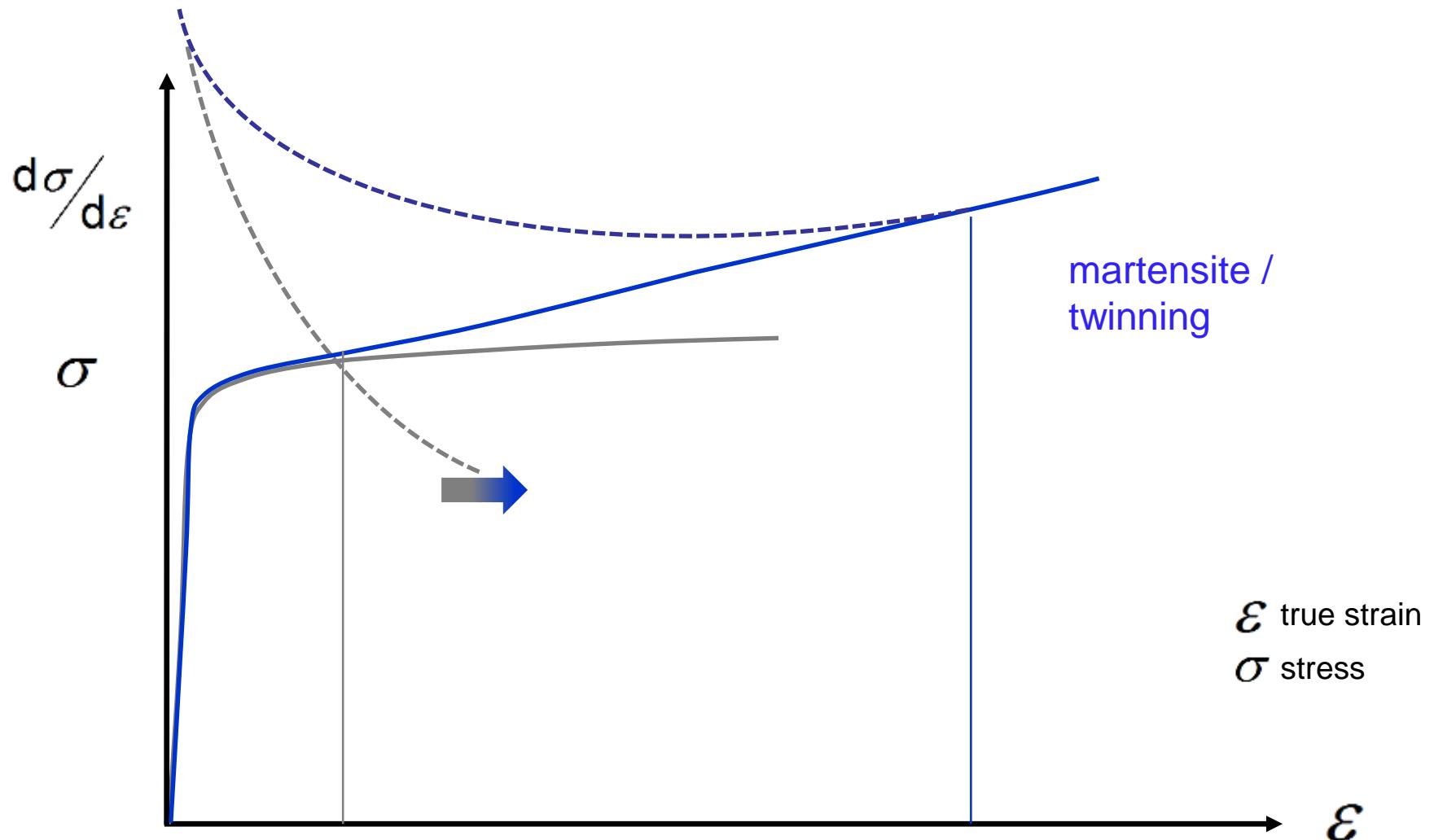
Dislocations and strain hardening

YEARS 1917-2017
100

Dislocations and strain hardening



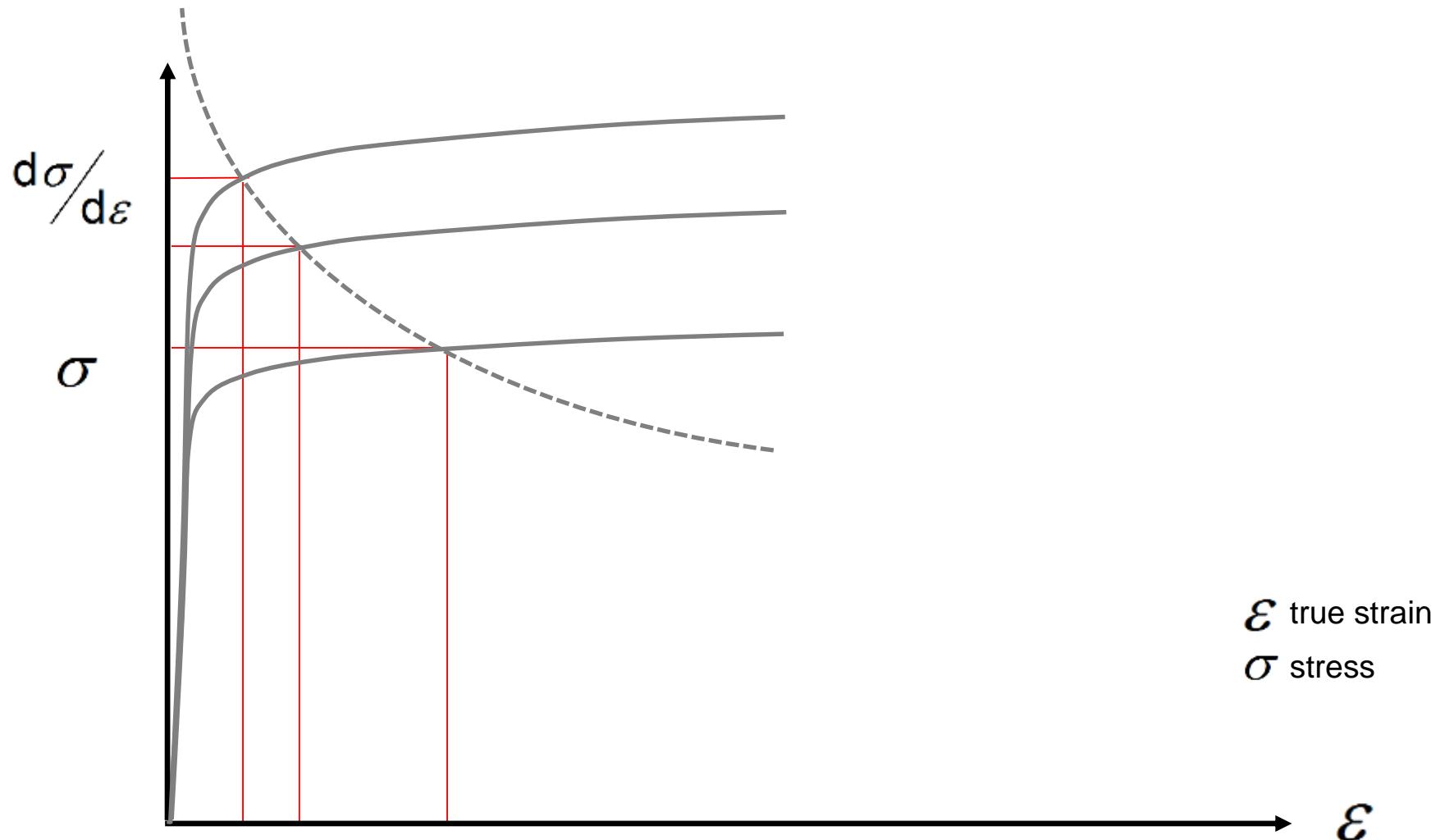
YEARS 1917–2017
100



Dislocations and strain hardening



YEARS 1917–2017
100

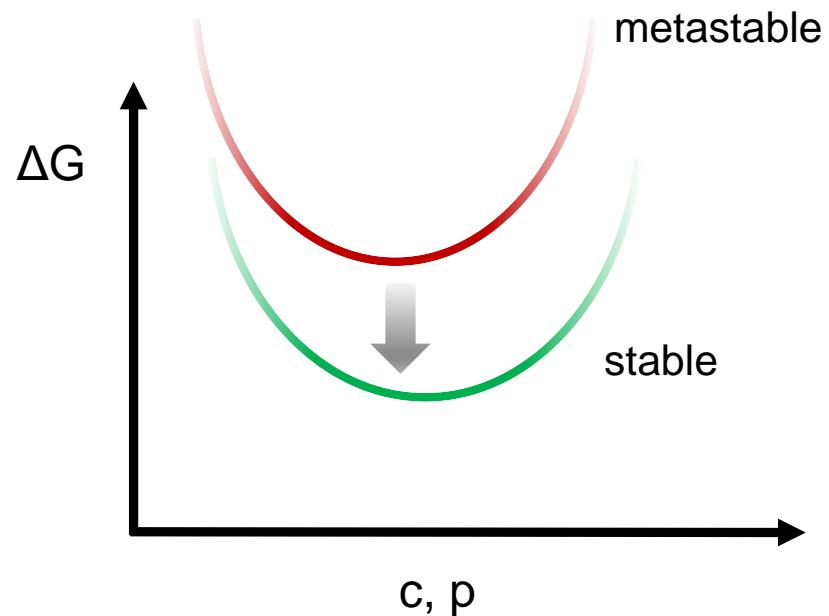
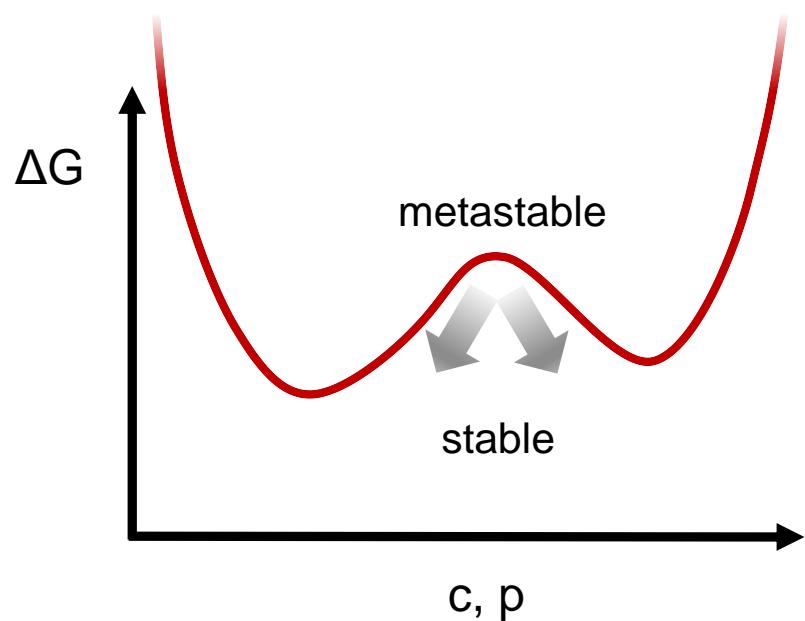


Metastability Alloy Design



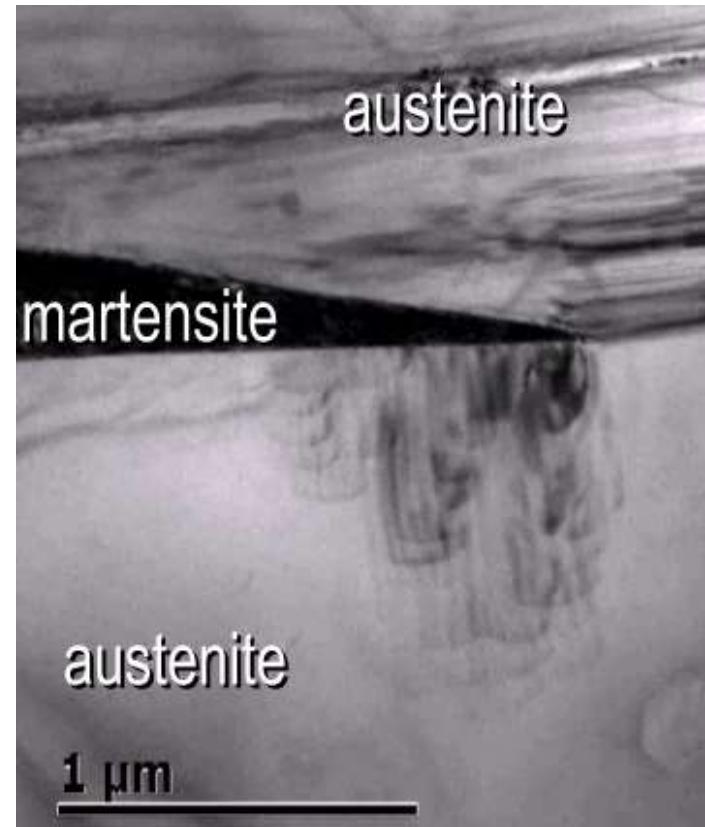
Mechanical Metastability

Chemical Metastability



Athermal transformations not affine,
not commensurate -
high misfit deformation

Multiple strain hardening effects

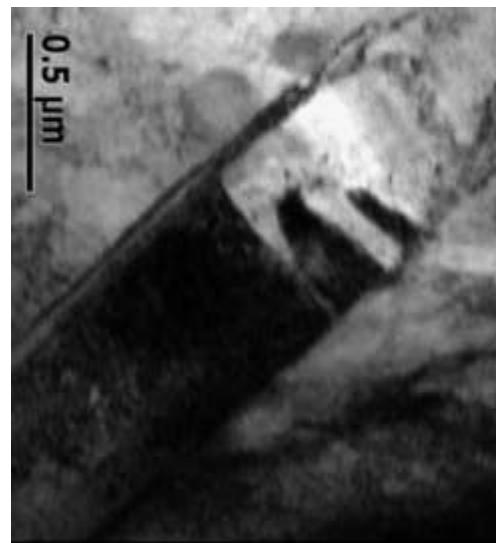
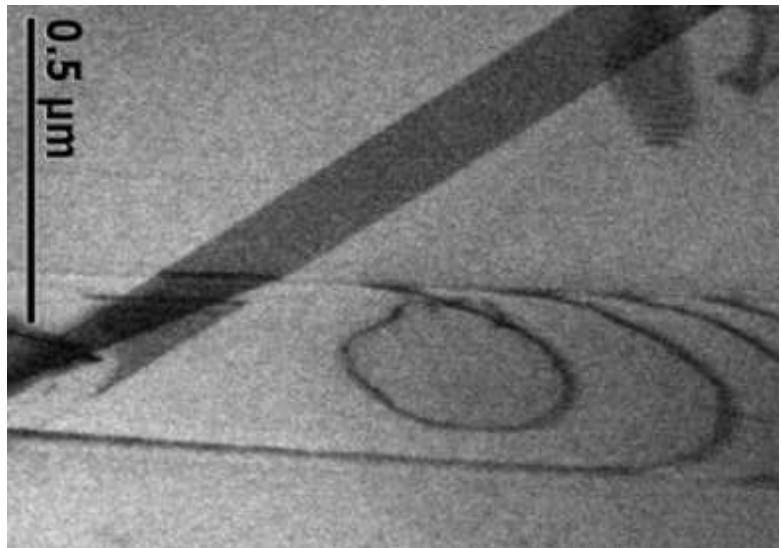


Confined at lattice defects

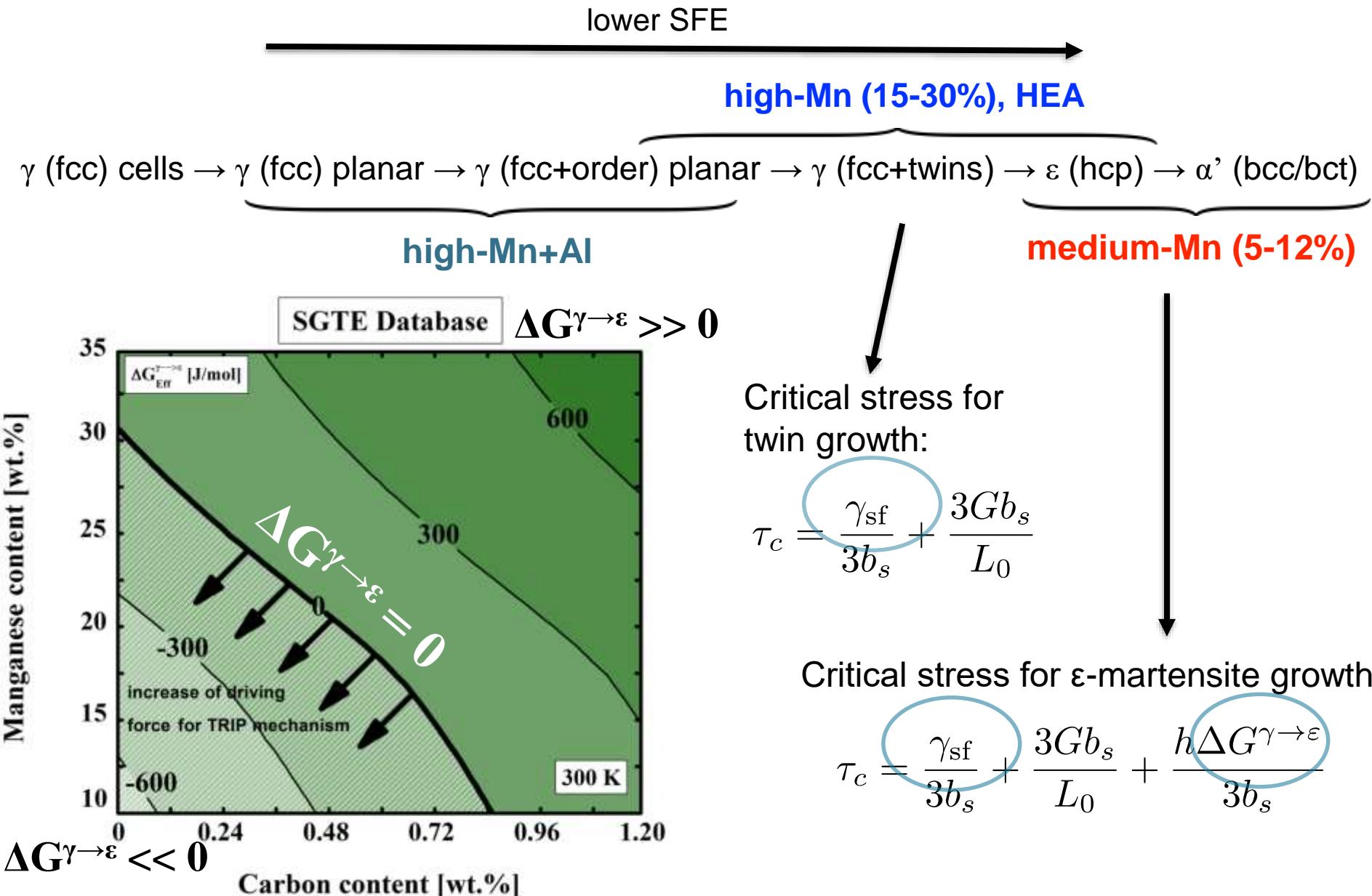
Chemical and structural size effects

Size effects in the bulk

- All for free and self-organized
- Thermodynamically tuned



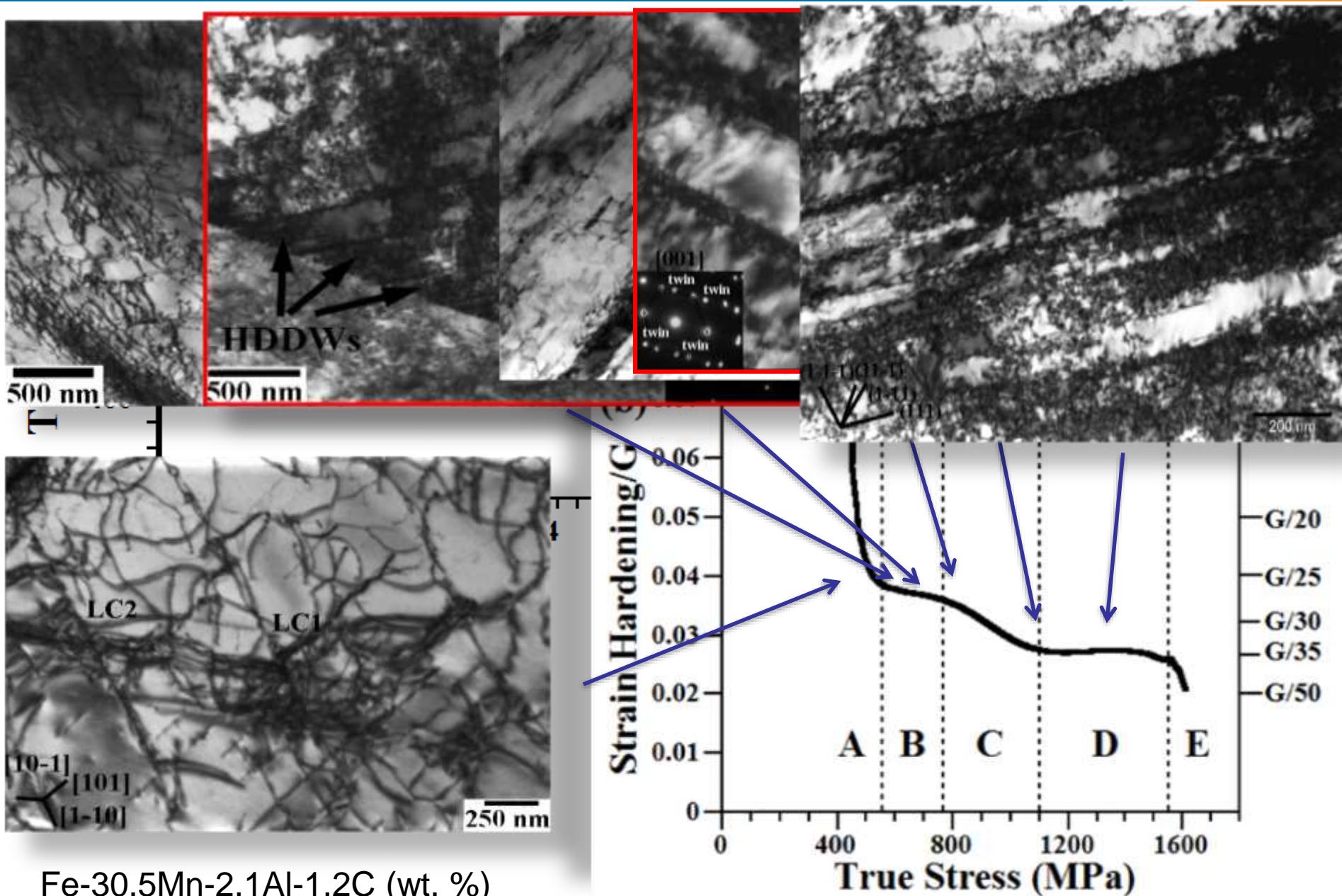
Role of the stacking fault energy

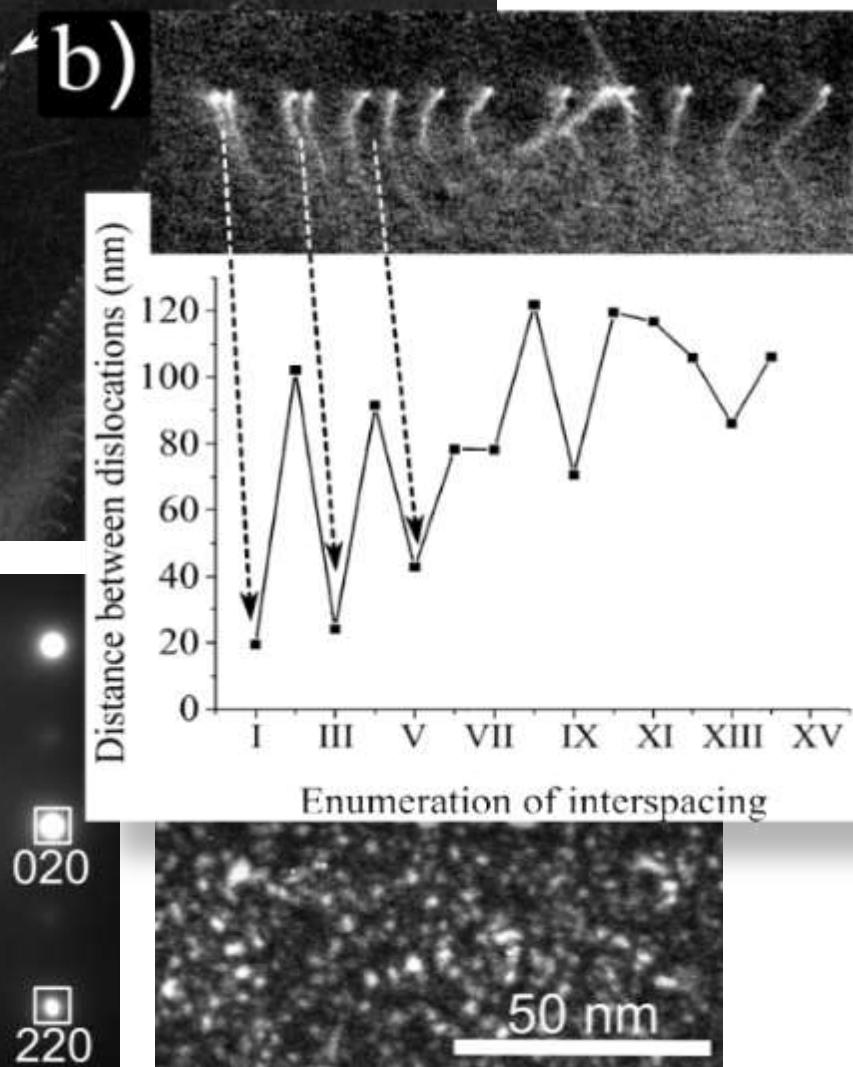
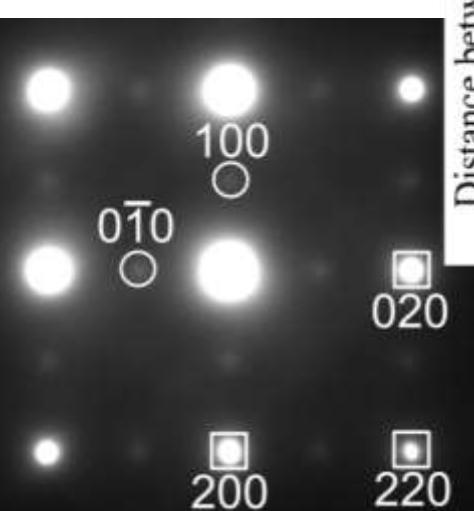
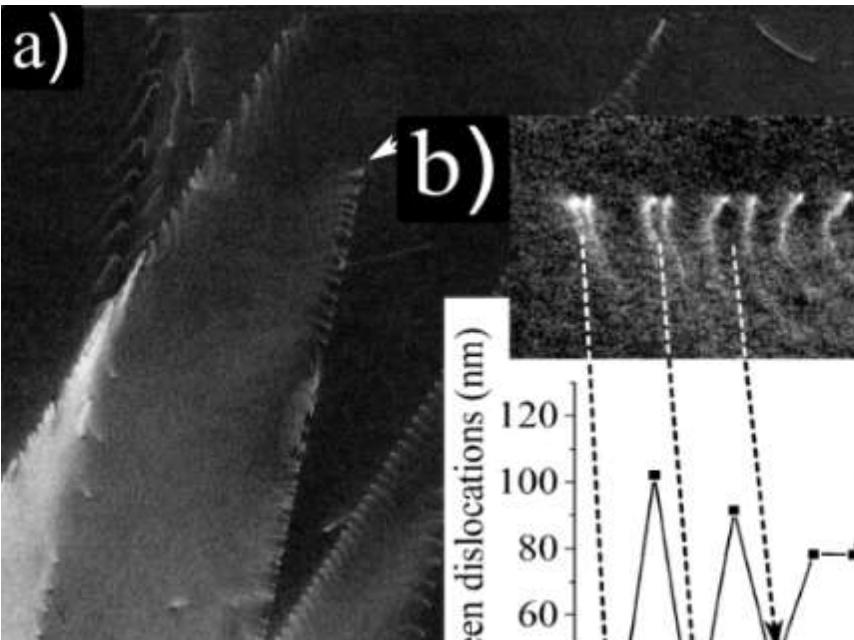


Fe-Mn-Al-C solid solution

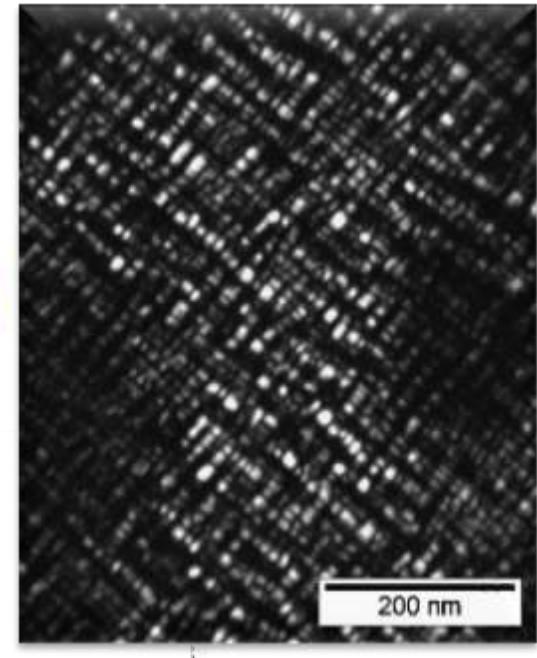
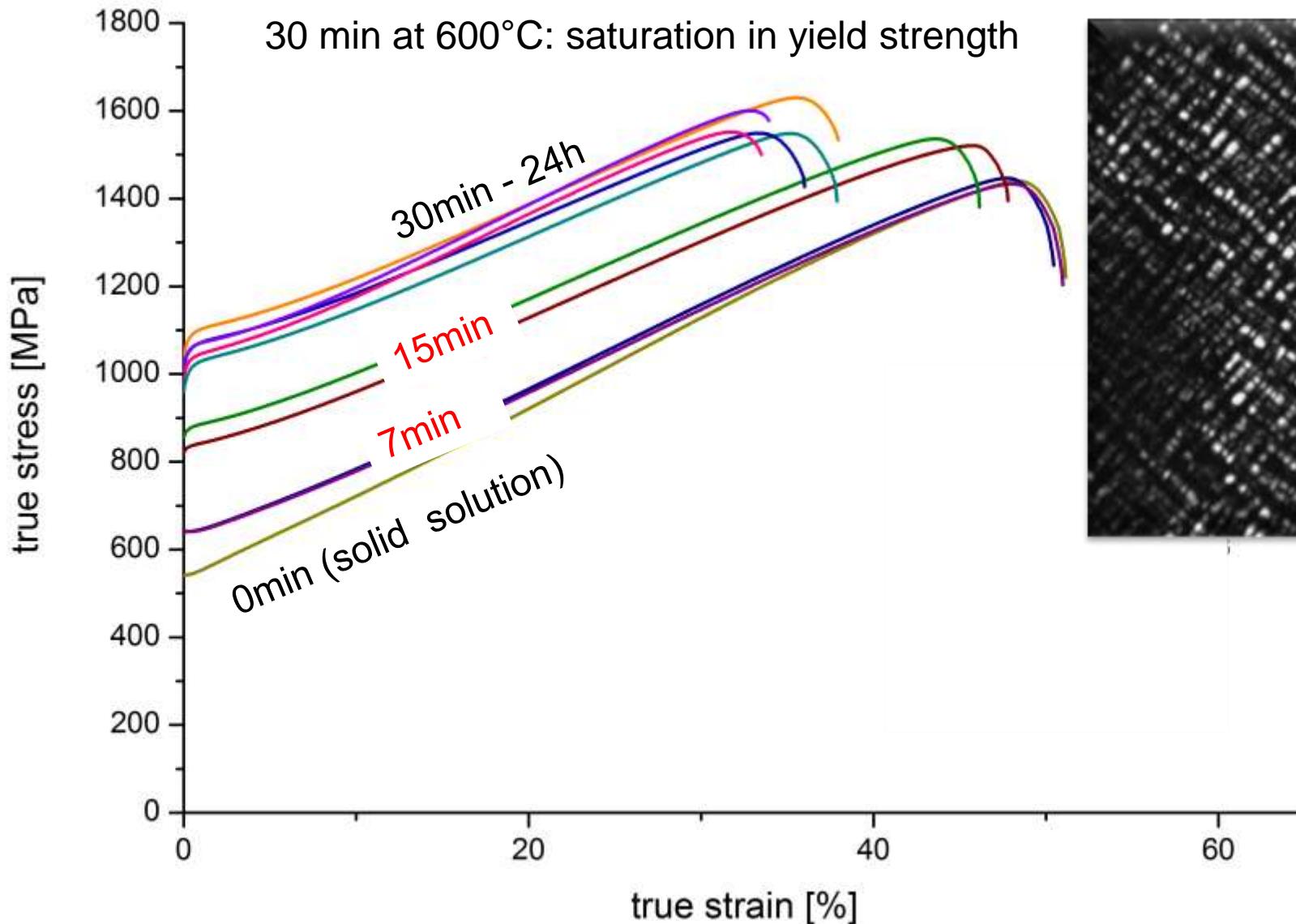


YEARS 1917-2017
100





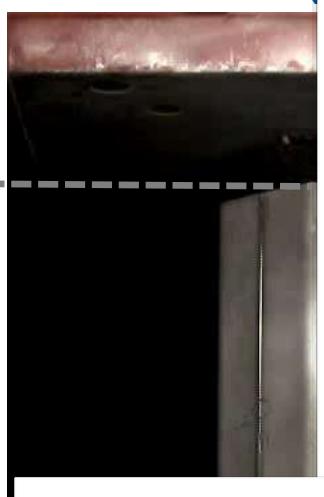
Fe-30Mn-8Al-1.2C, annealing at 600°C

YEARS 1917-2017
100

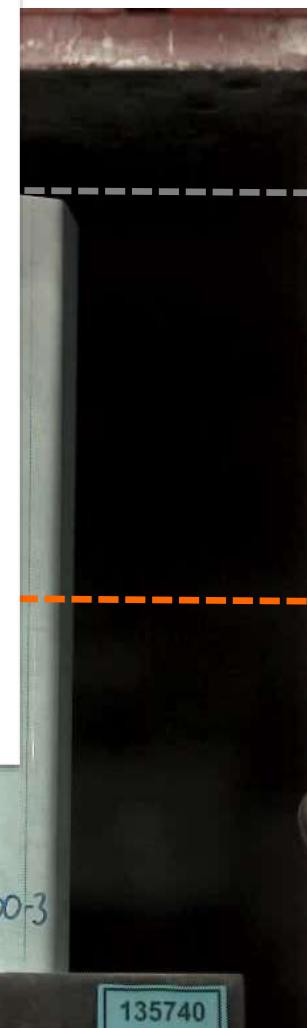
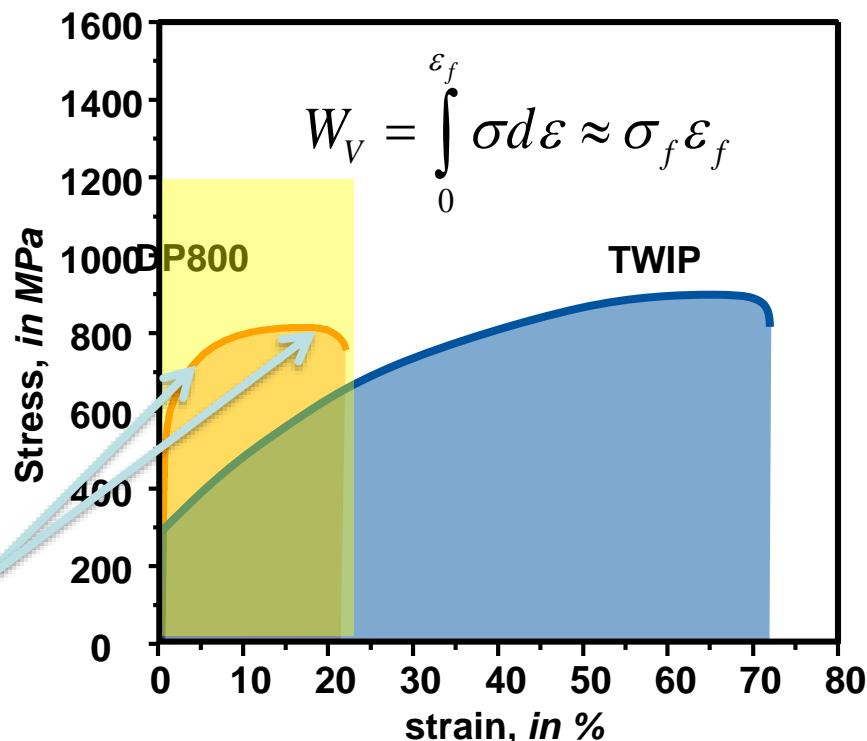
Strain rate 800/s: TWIP steel & DP800



TWIP (



Loading
points from
buckling



fka

135740

140

fka

DP800-3

135740

Role of the stacking fault energy



50 mJ/m^2

Stacking fault energy

$\cong 0 \text{ mJ/m}^2$

Dislocation slip

Stacking faults

Dislocation cell structures

Partial dislocations, reduced cross slip, planar slip

Chemical short range ordering and decomposition effects possible

Slip band formation

Dynamic slip band refinement

Mechanical twinning

Coexistence of fcc (γ), hcp (ε), bcc (α) / bct (α')

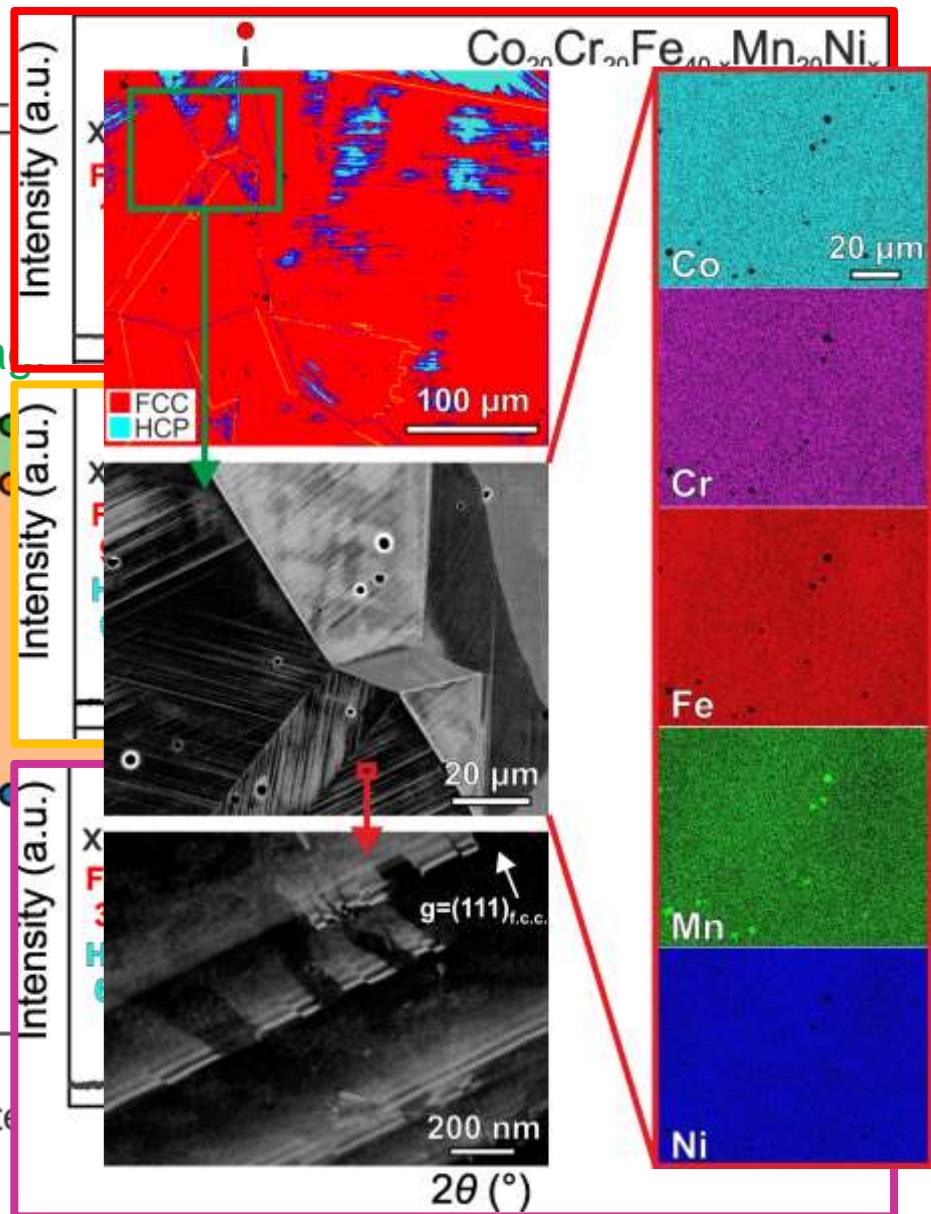
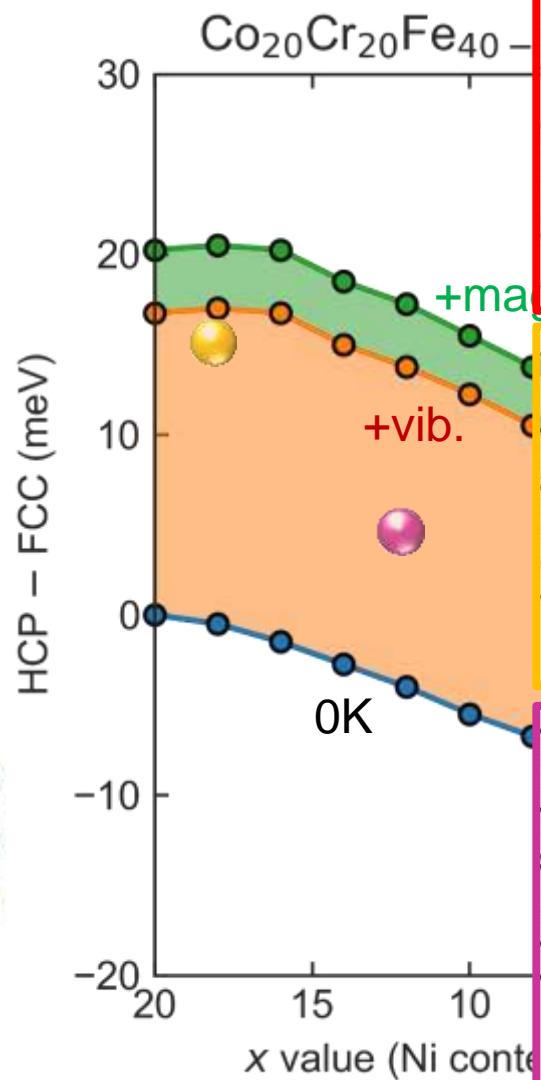
Less

More

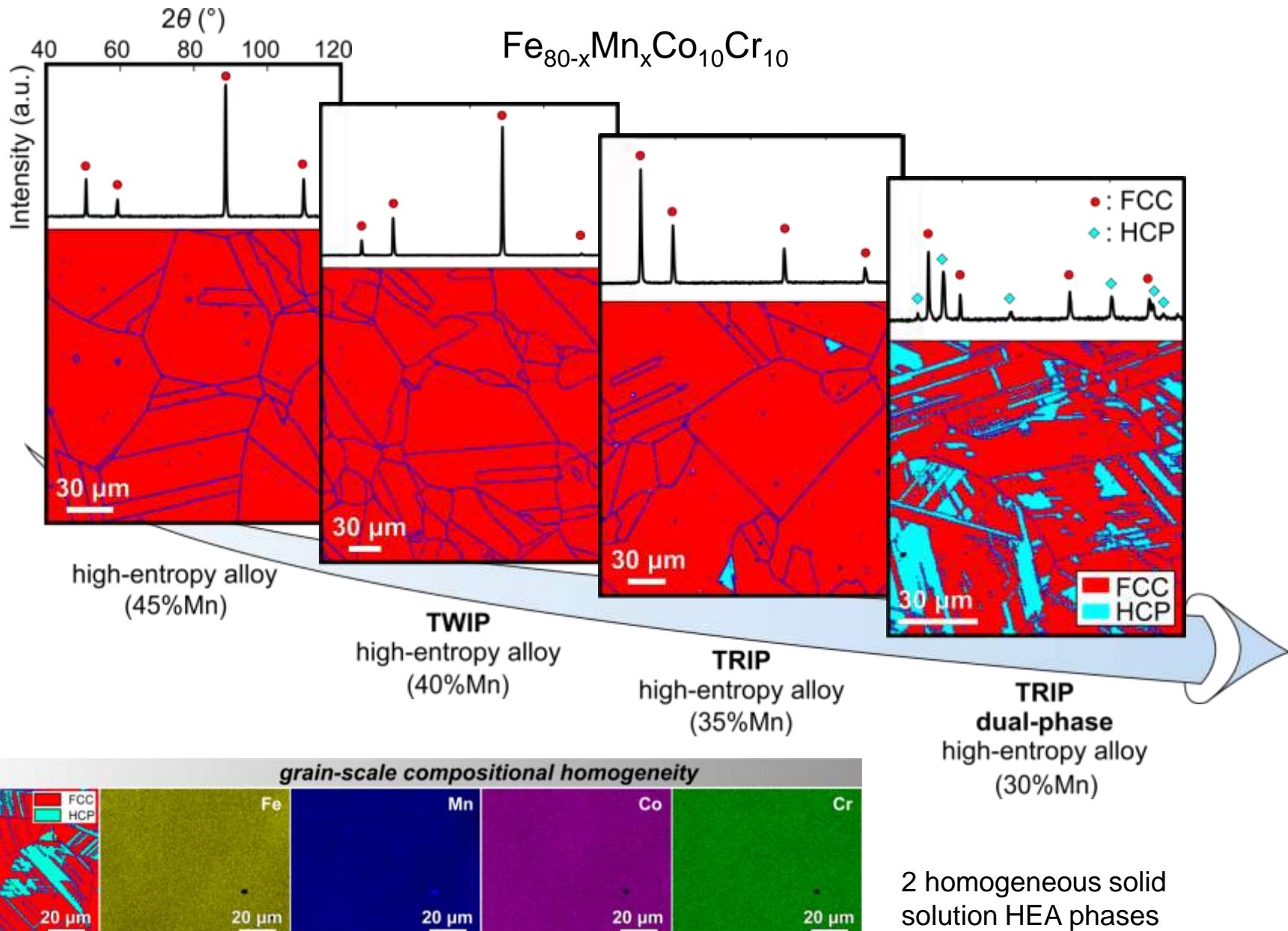
Unidirectional TRIP effect

Bidirectional TRIP effect

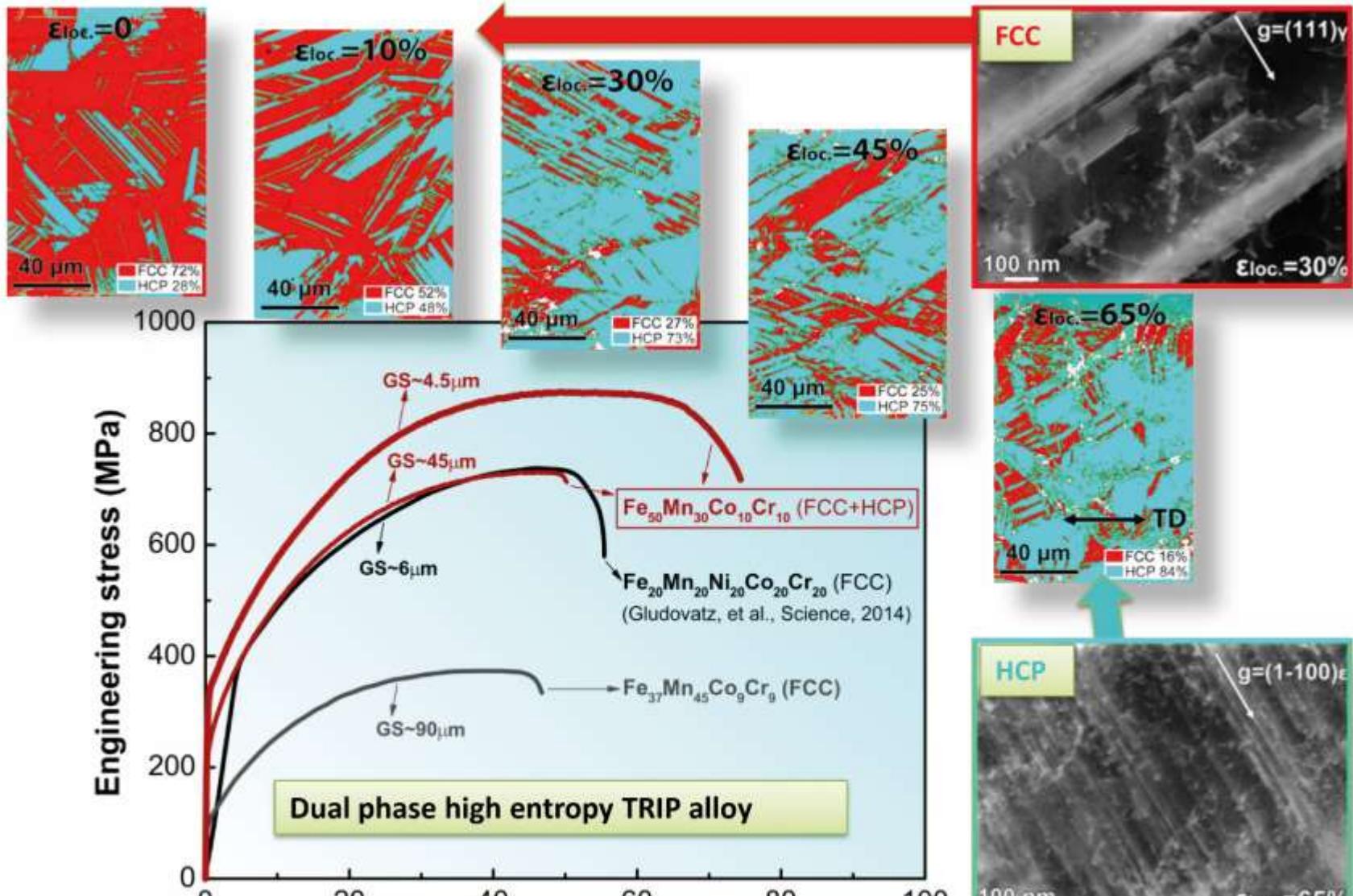
Dual phase high entropy alloy



Dual phase high entropy alloy



Dual phase high entropy alloy

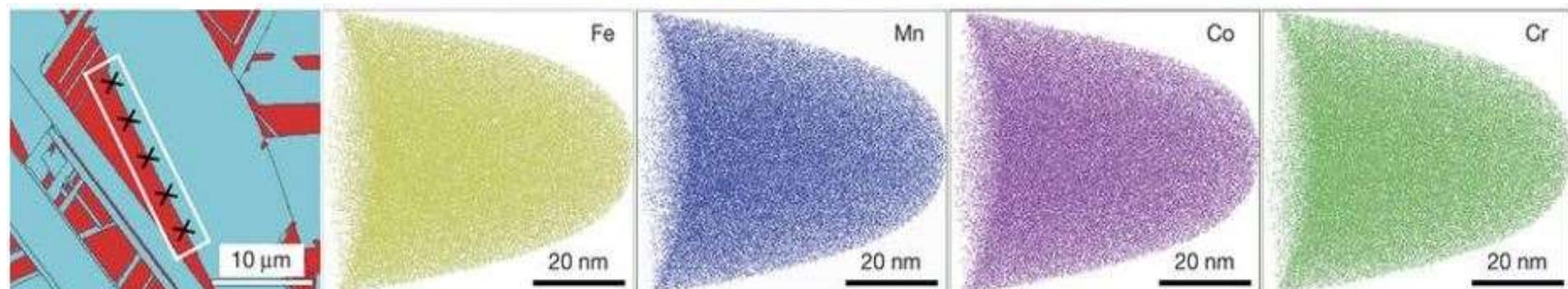
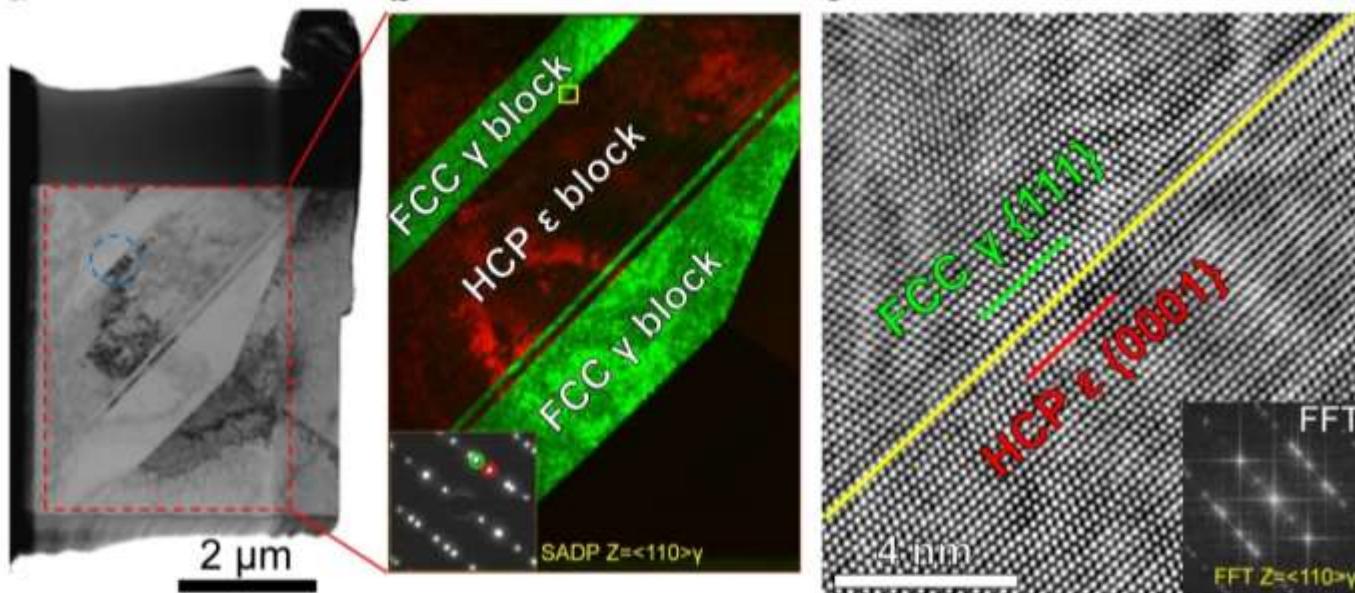


Z. Li et al. Nature 2016

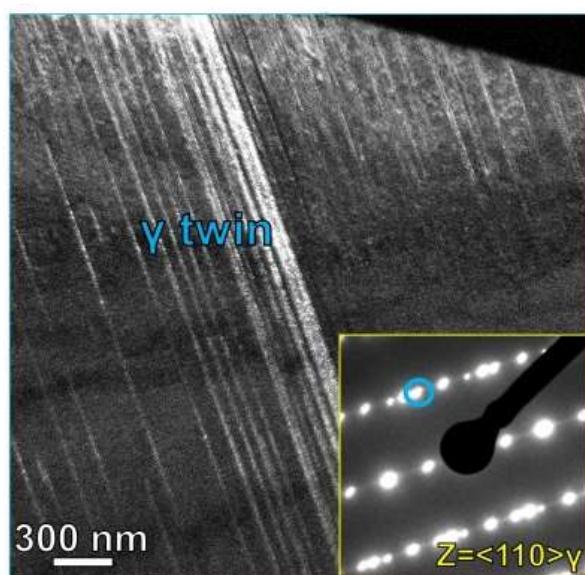
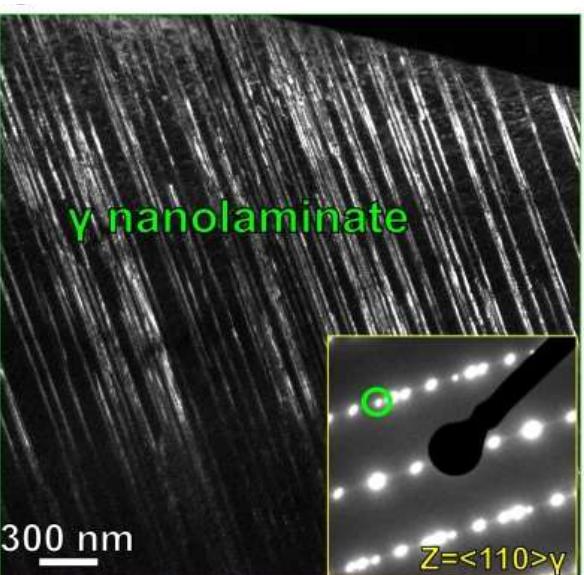
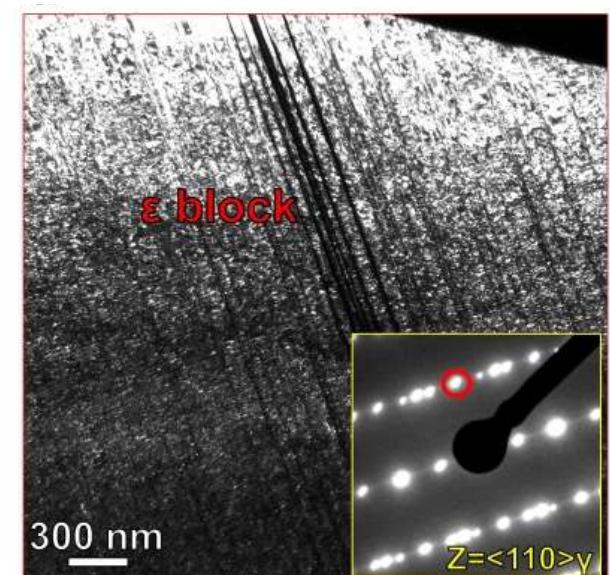
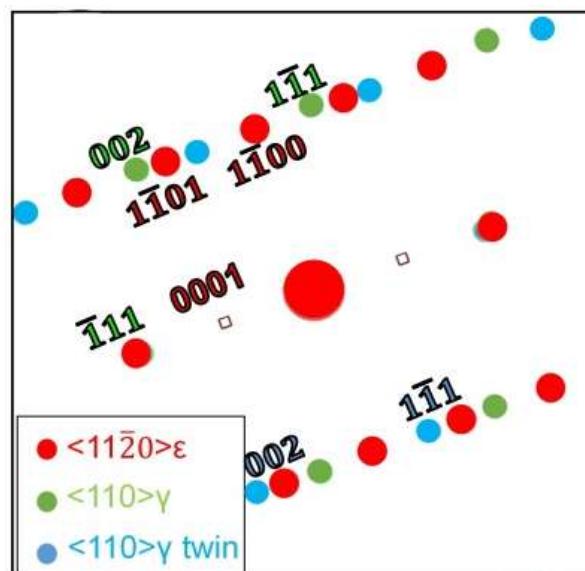
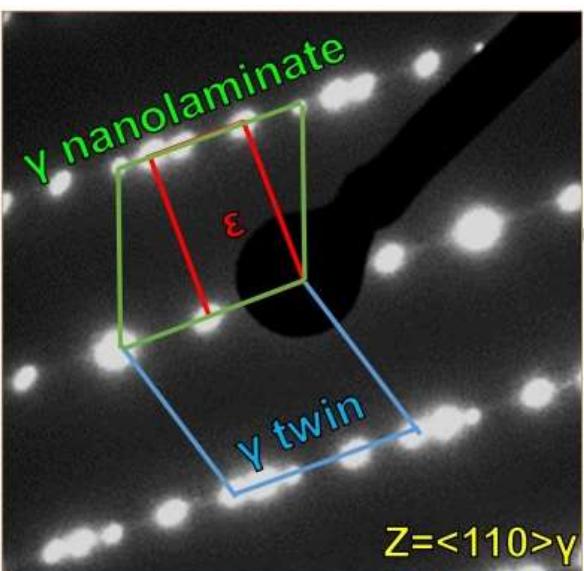
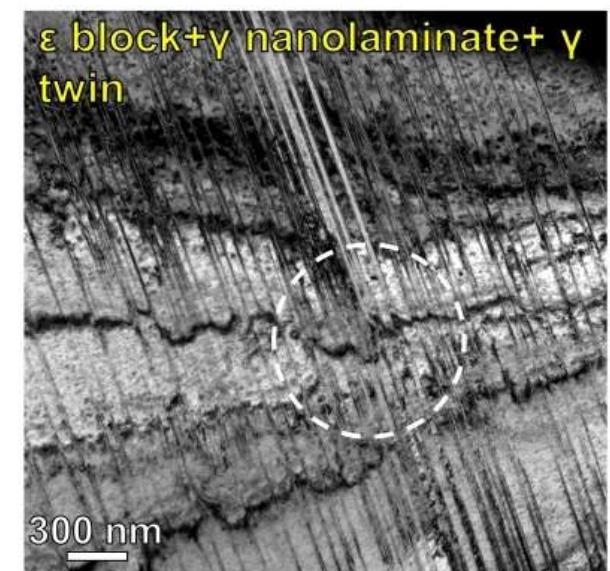
Interface between ϵ and γ (quenched)



- Fe50Mn30Co10Cr10
- STEM-EDS / APT / z-contrast: both phases chemically homogeneous
- SAED: $<11\bar{2}0>$ ϵ // $<110>$ γ ; {0001} ϵ // {111} γ

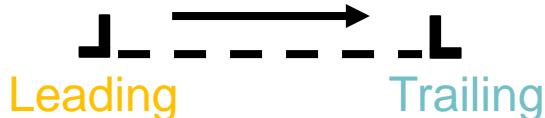


Nano- γ lamellae in ϵ under load

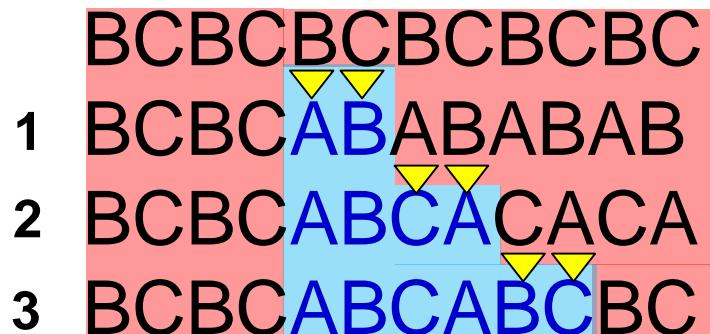


Transformation inside ε block ($\varepsilon \rightarrow \gamma$)

Process 1 (perfect FCC)



HCP \rightarrow Nano-FCC:

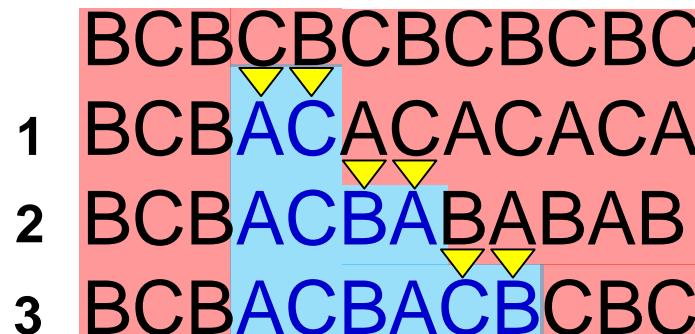


R. Bauer, et al. Philosophical Magazine, 2010.

Process 2 (nano-twin)

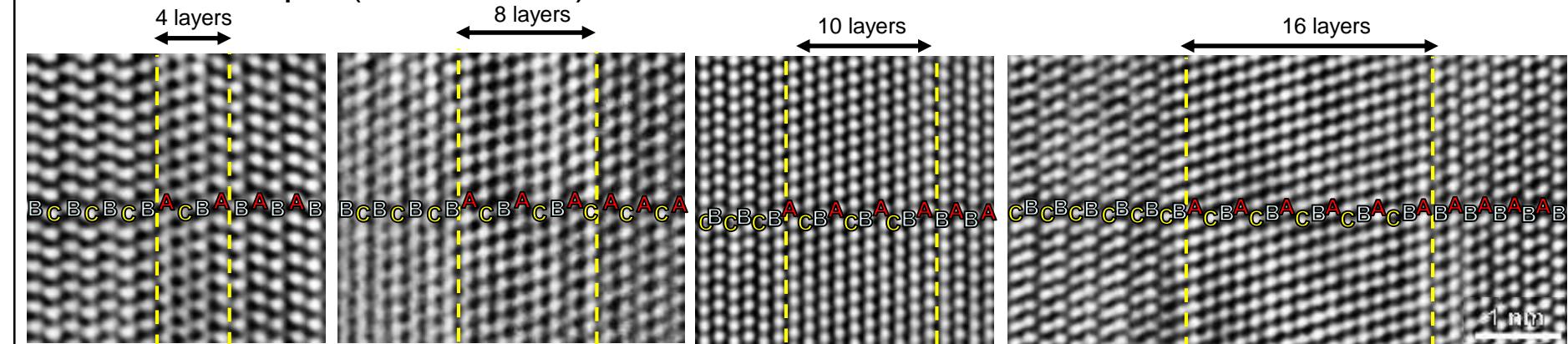


HCP \rightarrow Nano-FCC:



H.T. Wang, et al. Scripta Mater., 2018.

Deformed sample (twin relation)

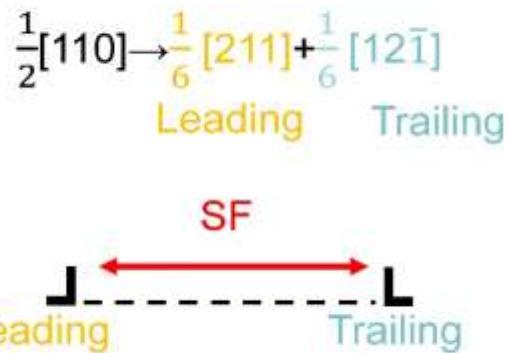


Transformation inside γ block ($\gamma \rightarrow \epsilon$)



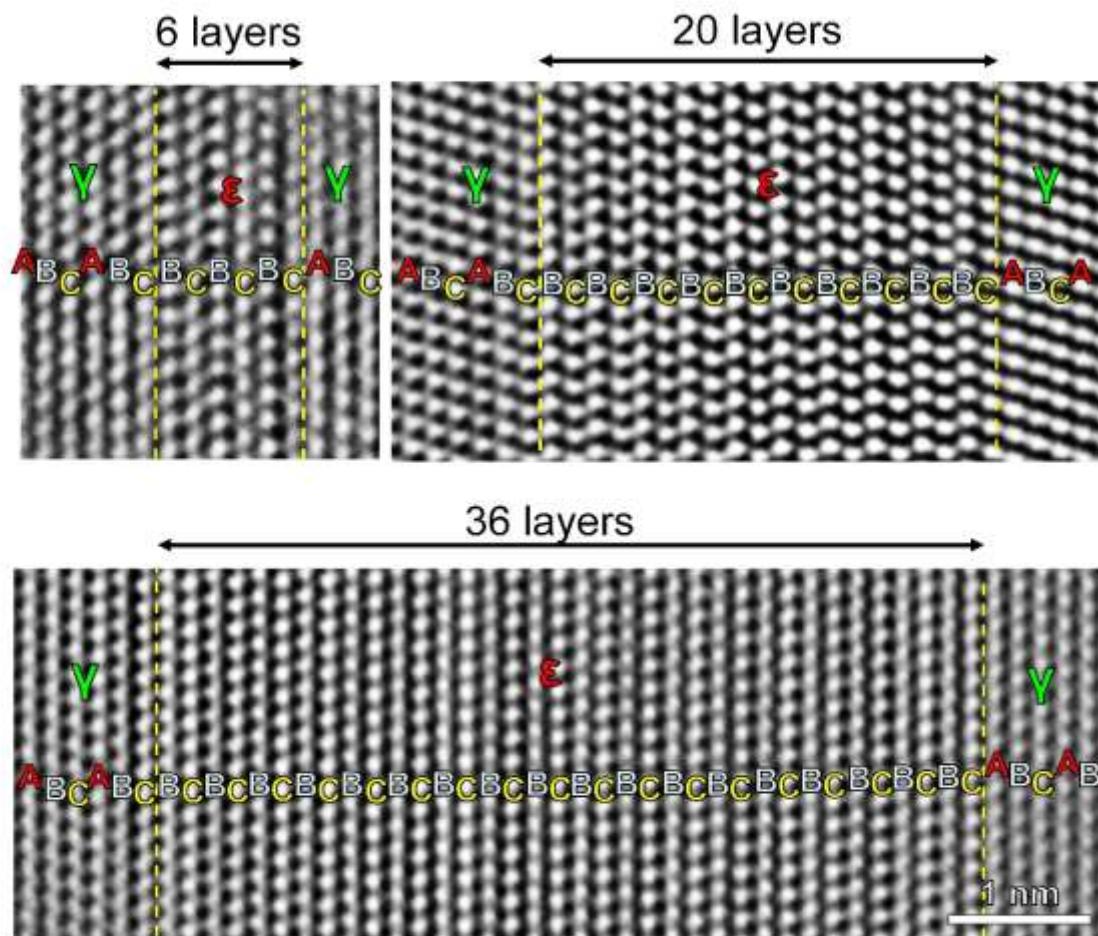
YEARS 1957–2017
100

Shockley partial dislocations



$\gamma \rightarrow \epsilon$:

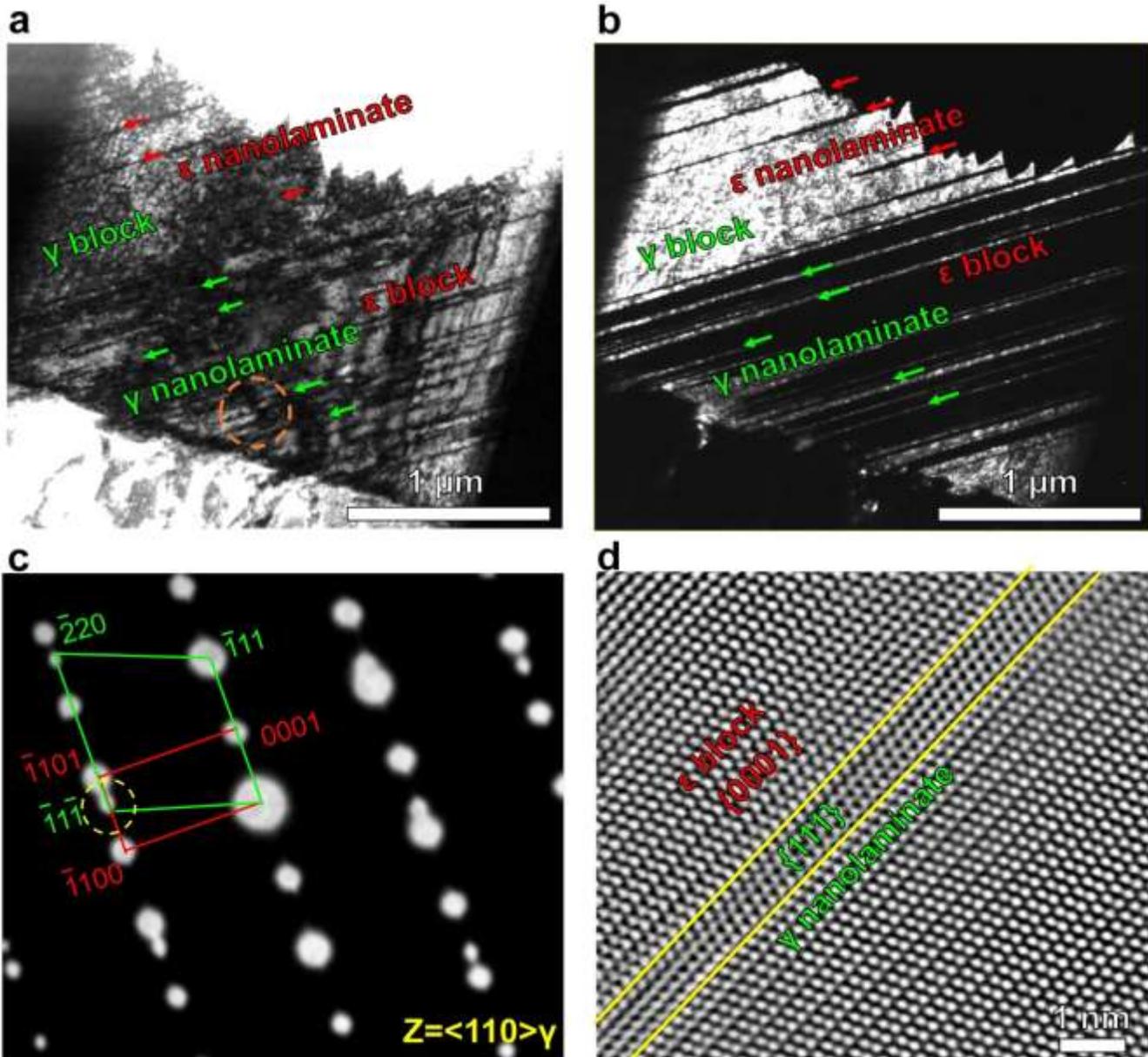
1 ABCABCABCABC
 1 ABCBCABCABC
 2 ABCBCBCABCAB
 3 ABCBCBCBCABC

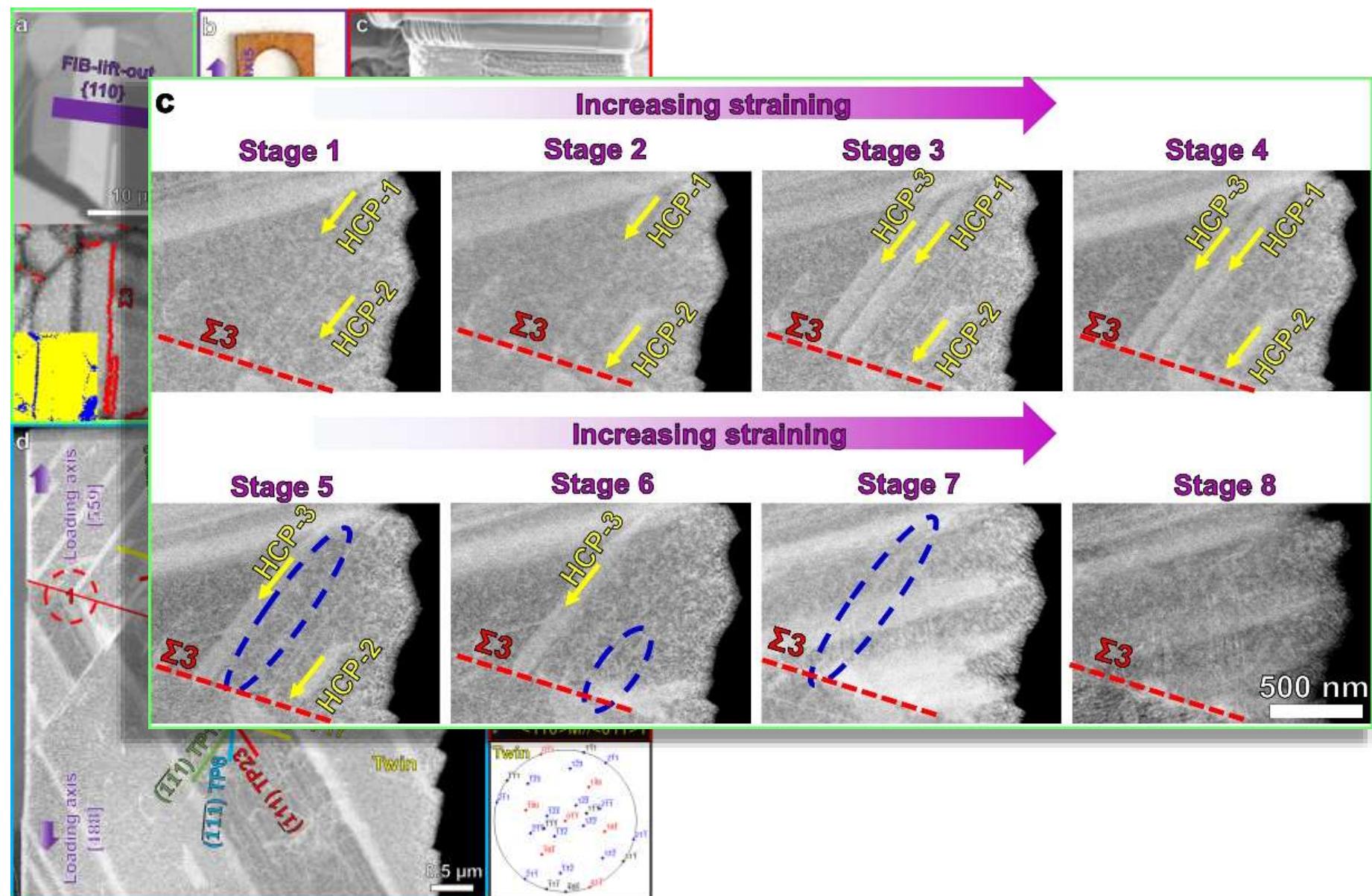


Nano-lamellae in ε and γ (quenched)



- Linear features in both the ε and γ blocks (DF-TEM)
- SFs (γ nano-lamellas): 87.6 ± 5.4 nm
- Thermodynamic stable at RT



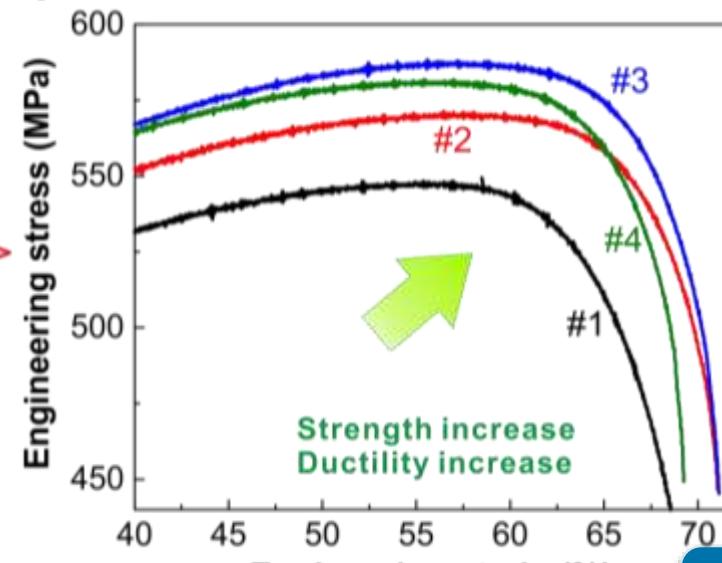
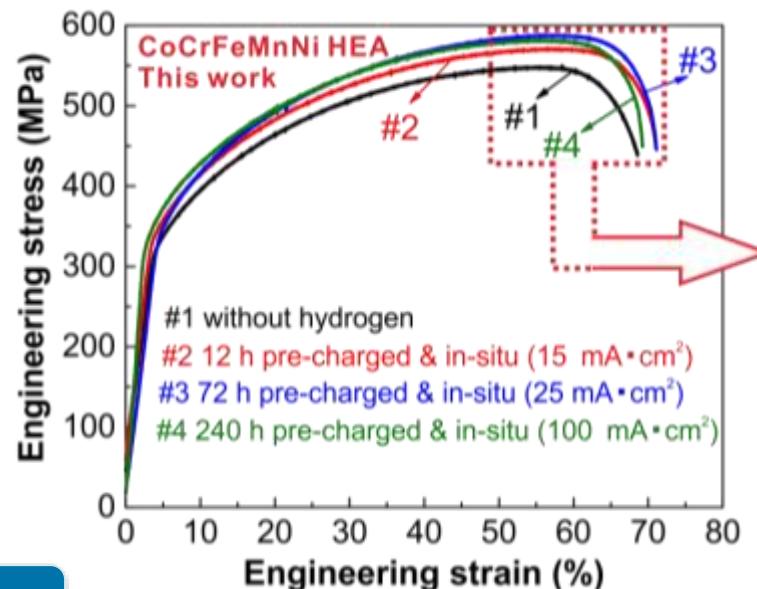


Effect of Hydrogen: equim.-FeNiCrMnCo

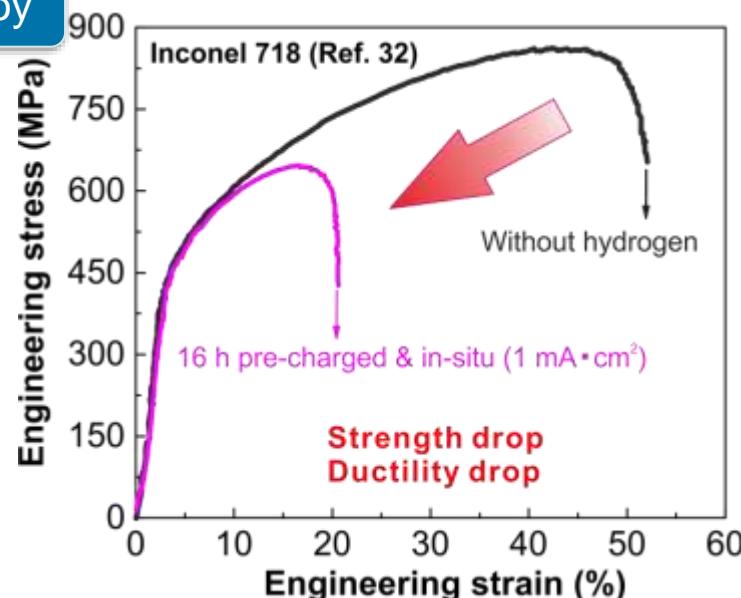


YEARS 1957-2017
100

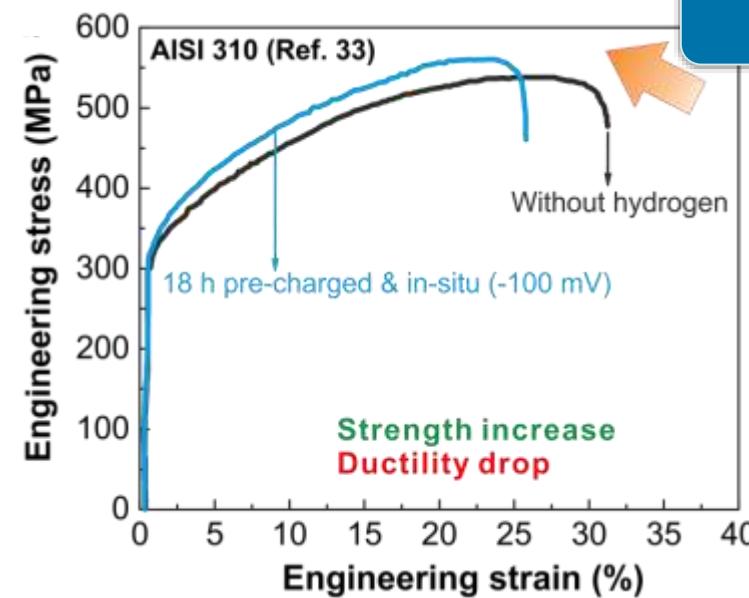
HEA



Superalloy



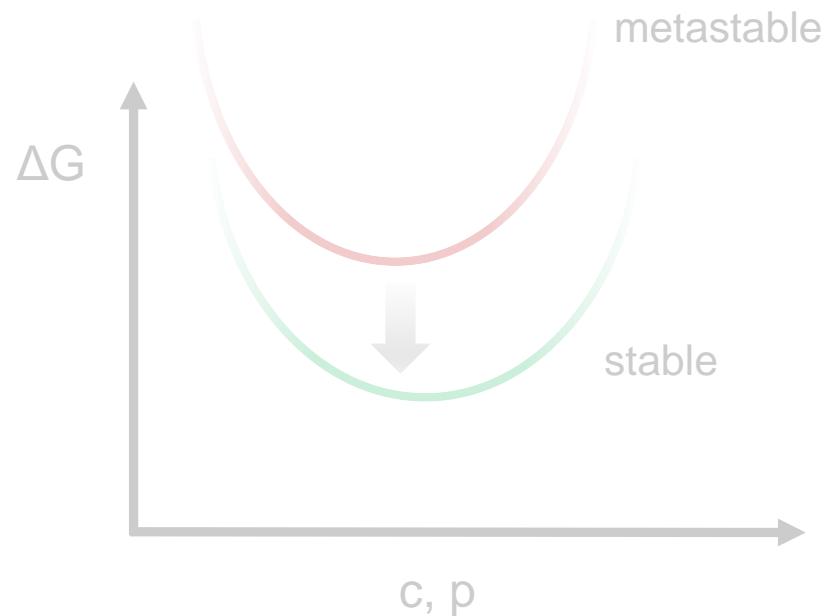
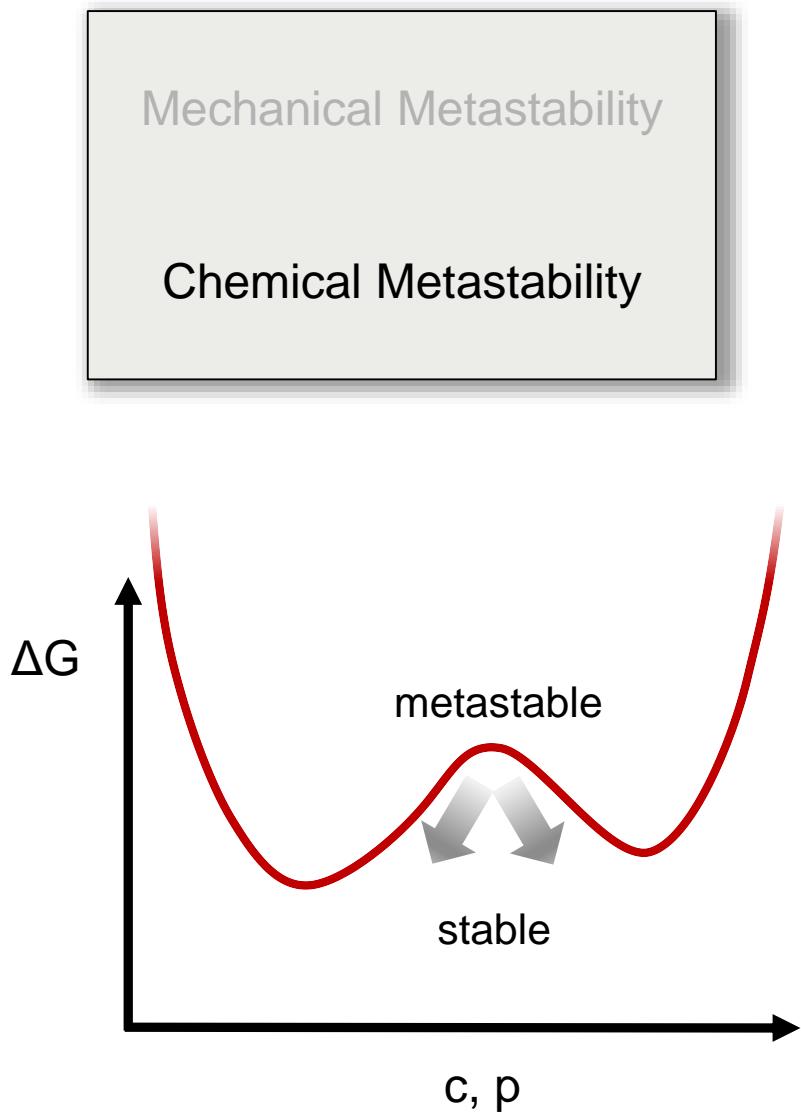
Stainless steel



56Ni-18Fe-18Cr-5Nb-3Mo

55Fe-25Cr-20Ni

Metastability Alloy Design

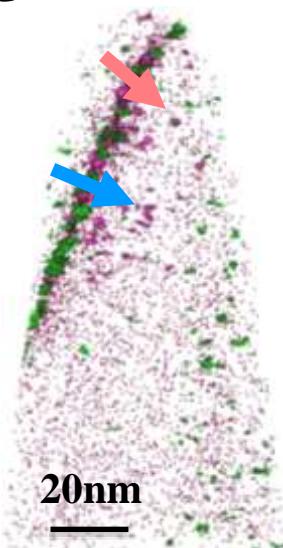


Interface co-segregation in FeMnCrNiCo

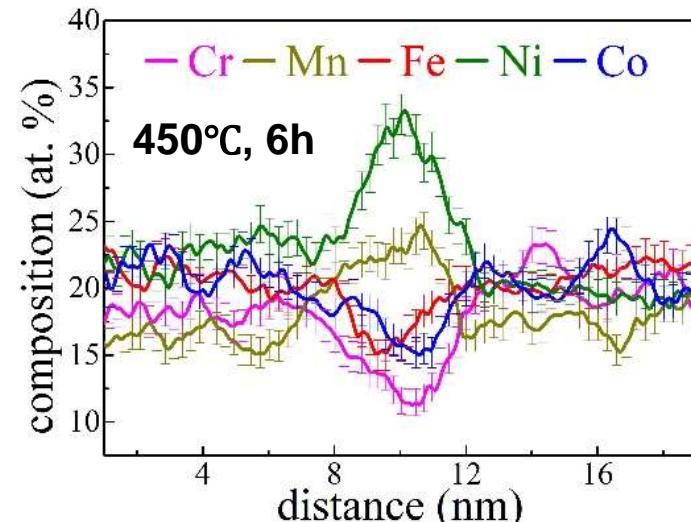
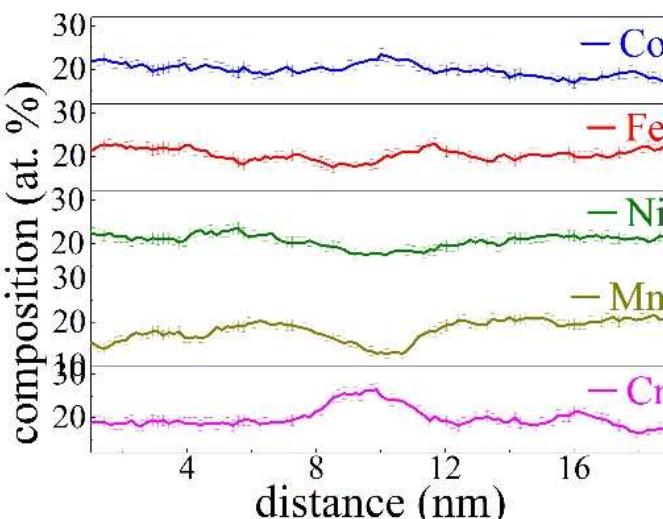


450°C

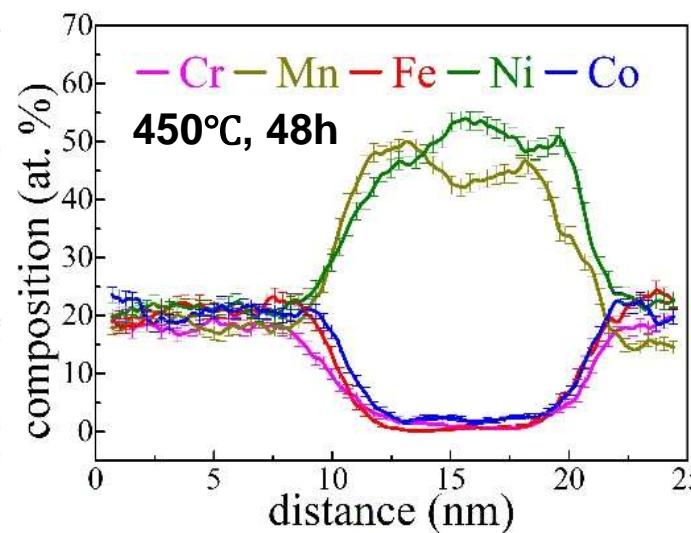
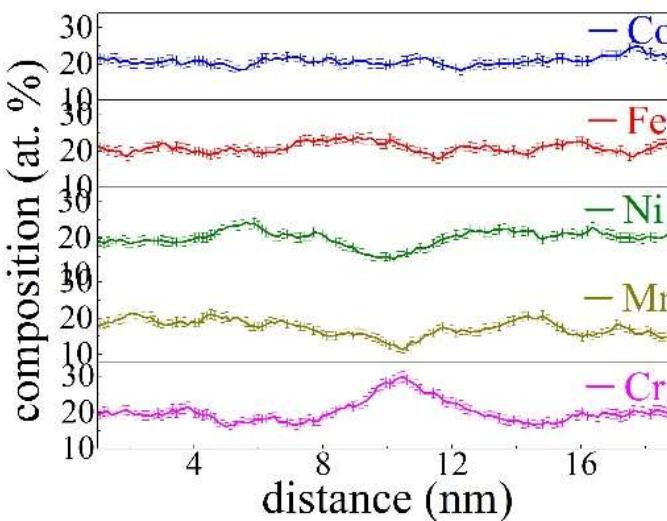
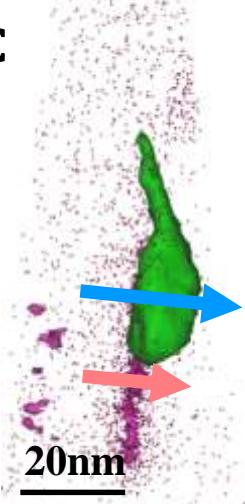
6h



Analysis perpendicular to grain boundary



450°C
48h



GB co-segregation in Cantor HEA

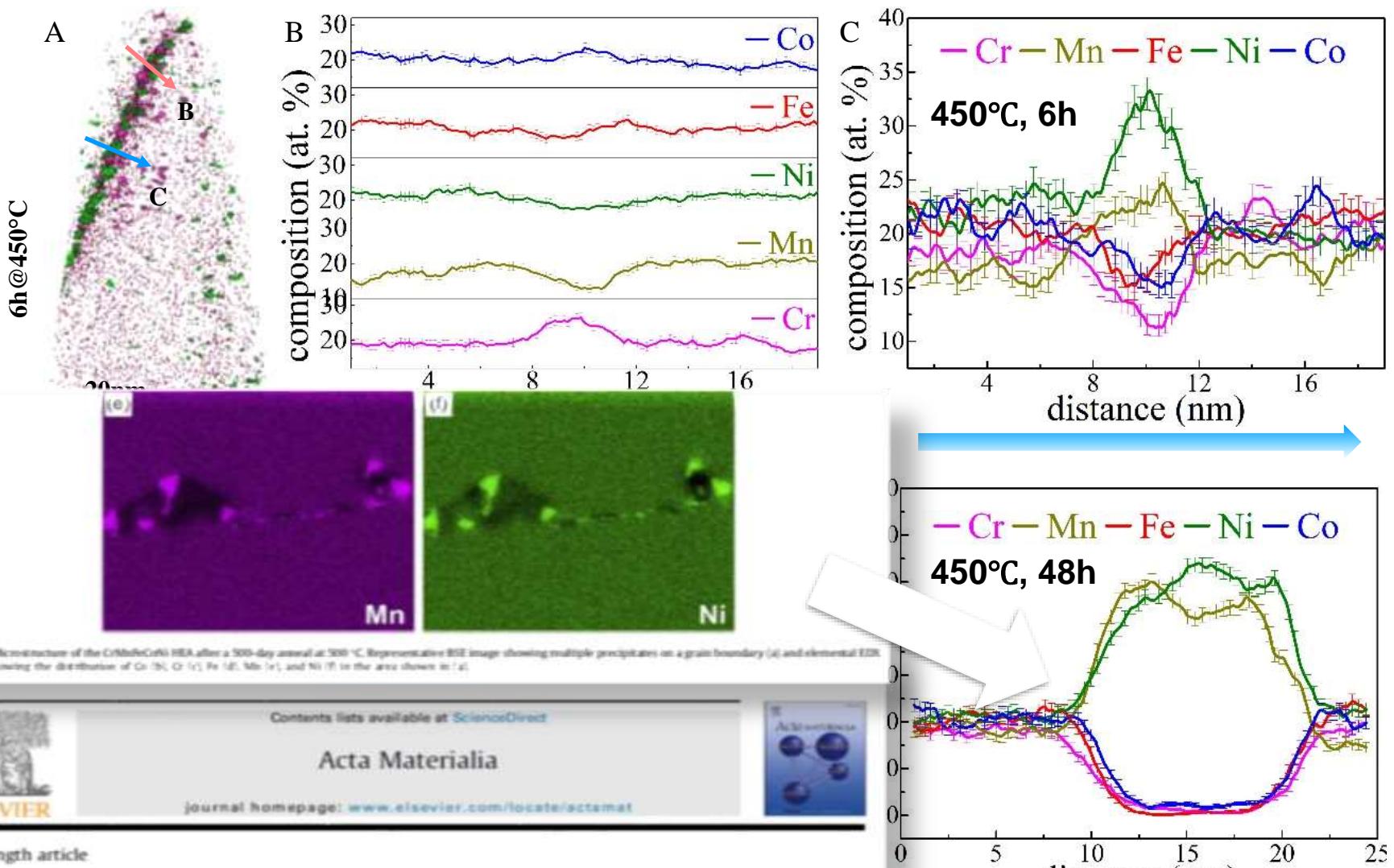


Fig. 2. Microstructure of the CrMnFeCoNi HEA after a 500-day anneal at 500 °C. Representative ESE image showing multiple precipitates on a grain boundary (a) and elemental EDS maps showing the distribution of Cr (b), Cr (c), Fe (d), Mn (e), and Ni (f) in the area shown in (a).



Contents lists available at ScienceDirect

Acta Materialia

journal homepage: www.elsevier.com/locate/actamat



Full length article

Decomposition of the single-phase high-entropy alloy CrMnFeCoNi after prolonged anneals at intermediate temperatures



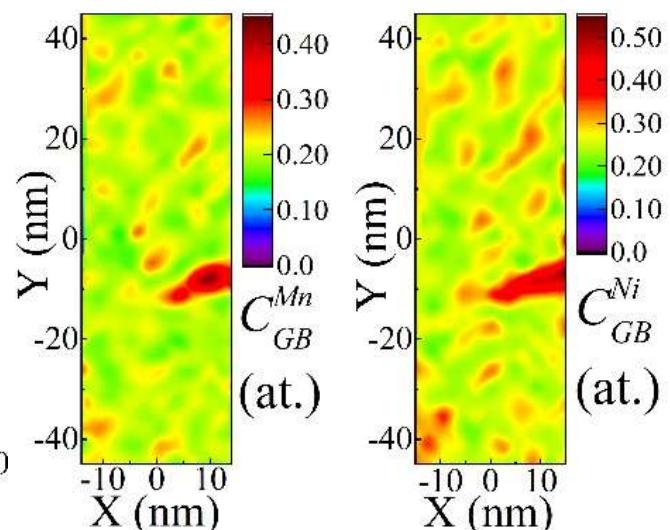
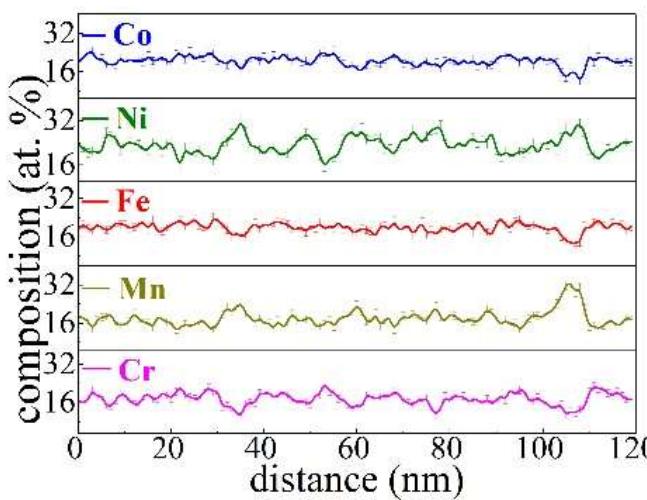
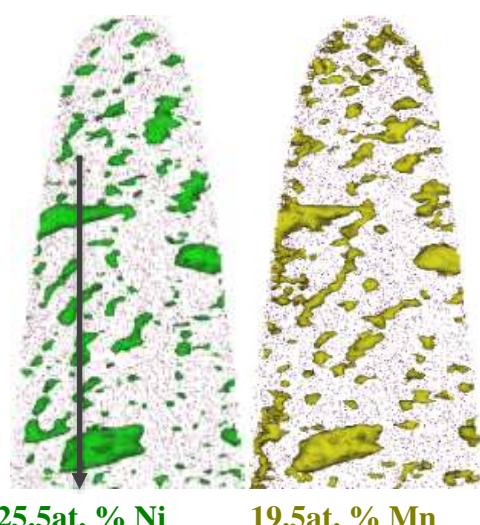
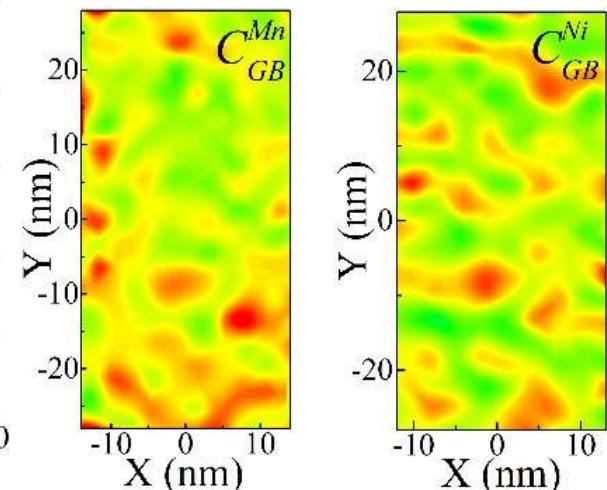
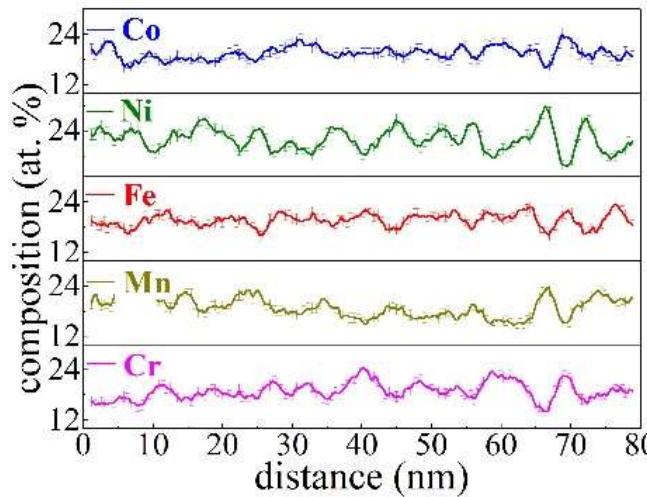
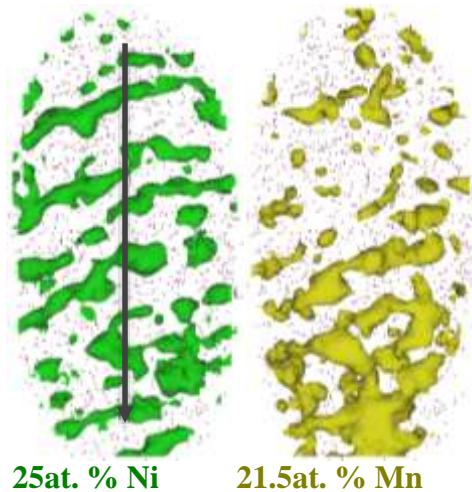
F. Otto^a, A. Dlouhý^b, K.G. Pradeep^c, M. Kuběnová^b, D. Raabe^c, G. Eggeler^a, E.P. George^{a,*}

Interface co-segregation in FeMnCrNiCo



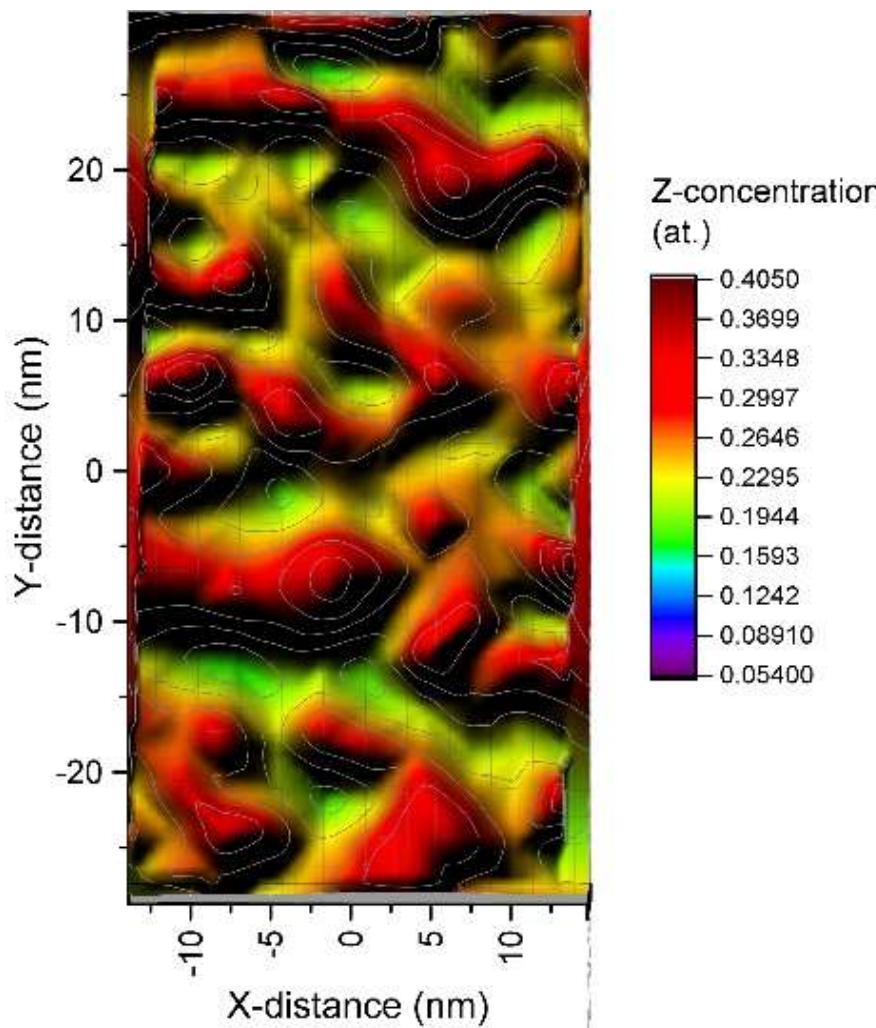
450°C, 6h

Analysis inside grain boundary plane

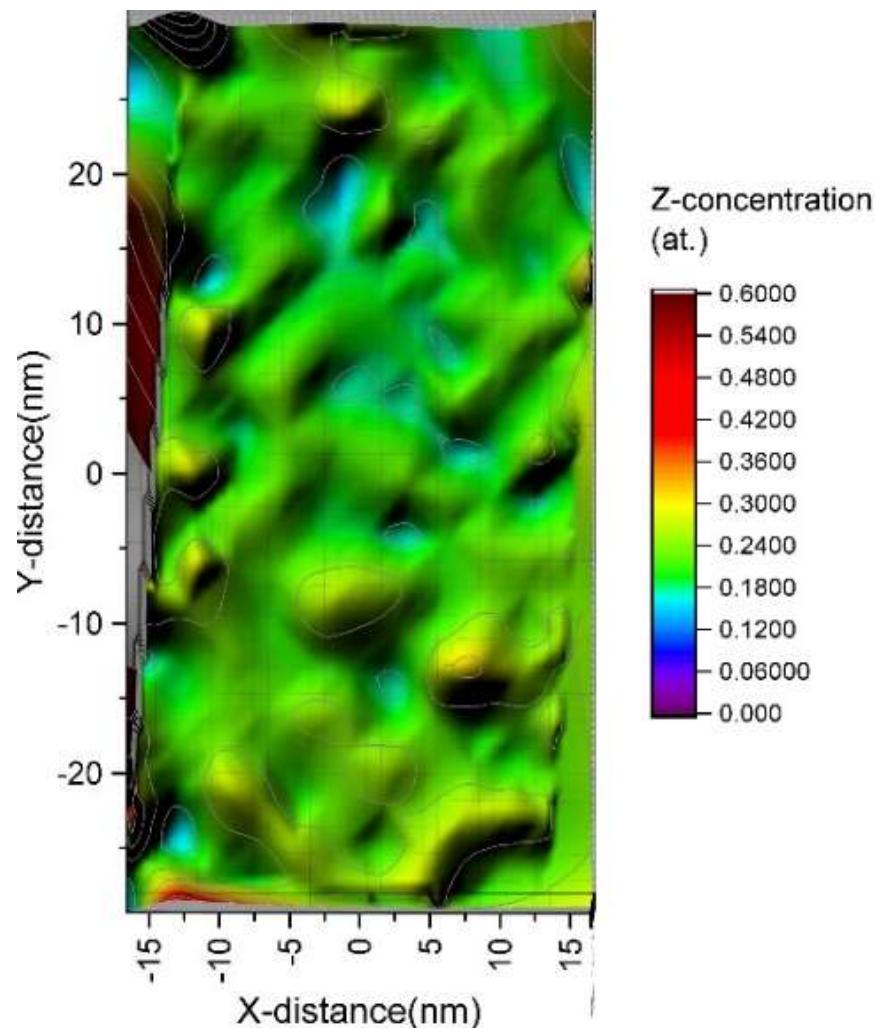


450°C, 18h

Interface co-segregation in FeMnCrNiCo



Ni in GB plane, 450°C, 6h



Mn in GB plane, 450°C, 6h

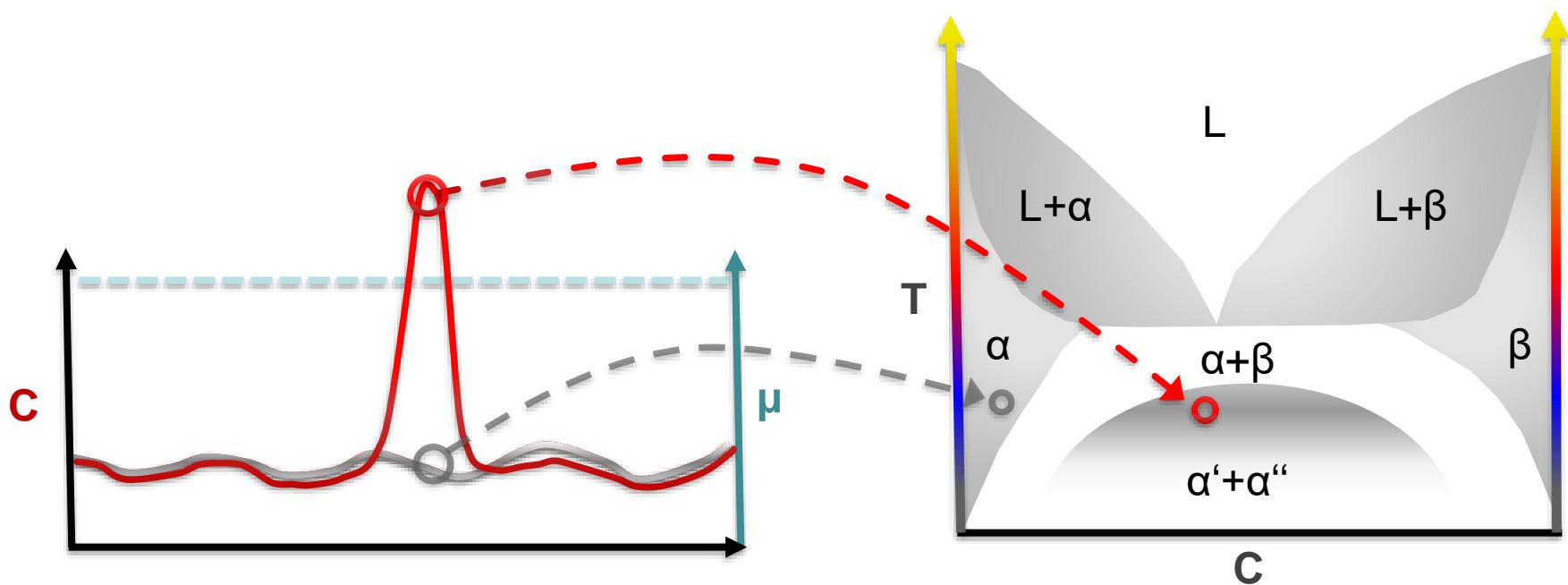
Defect decoration & thermodynamics



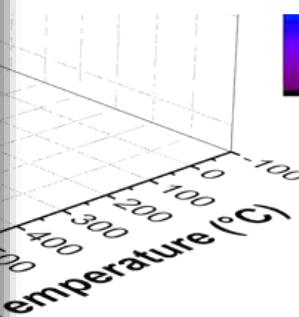
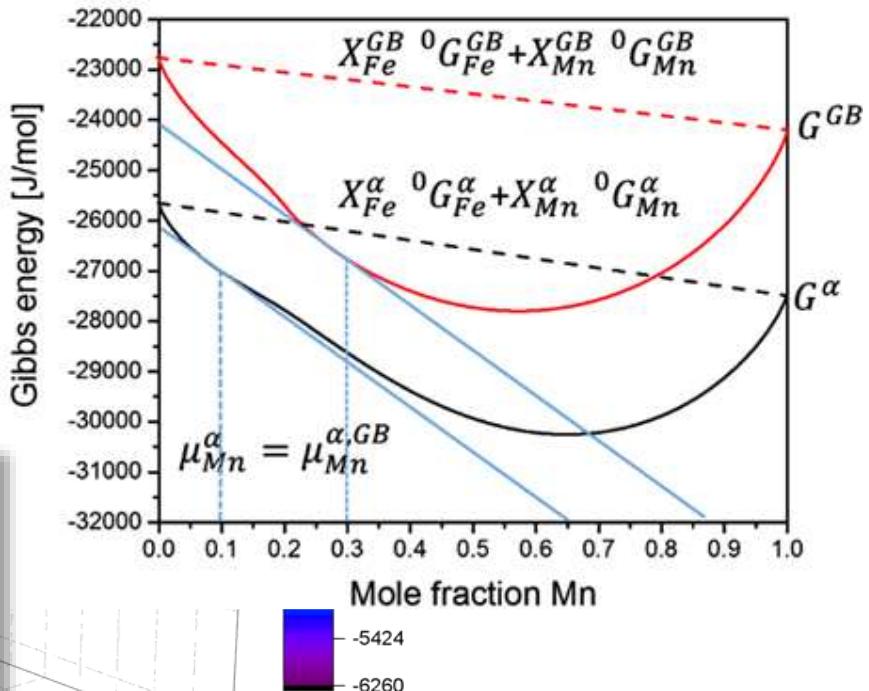
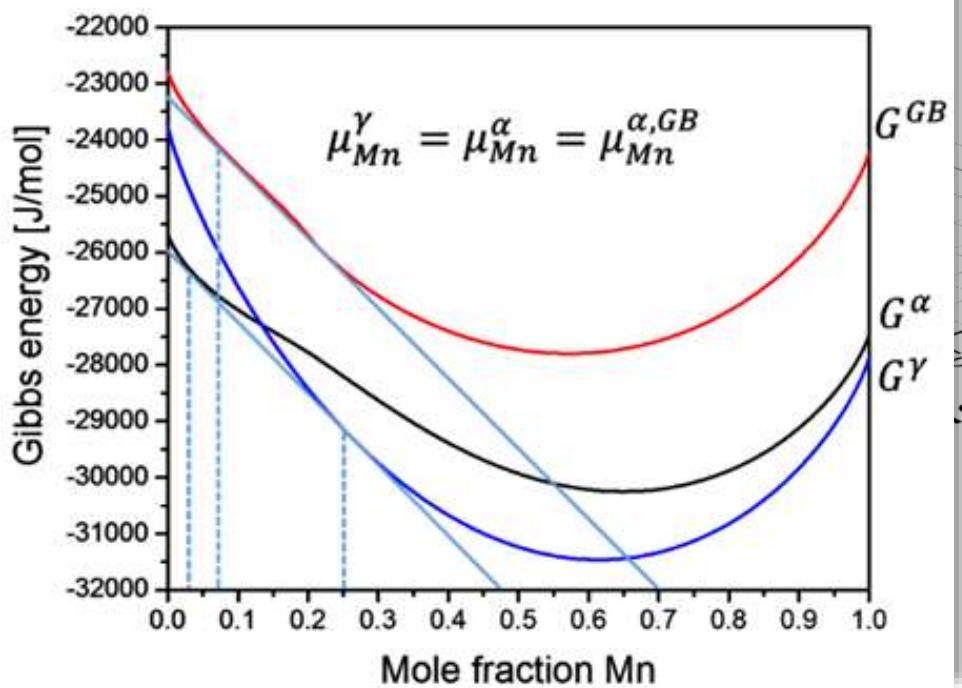
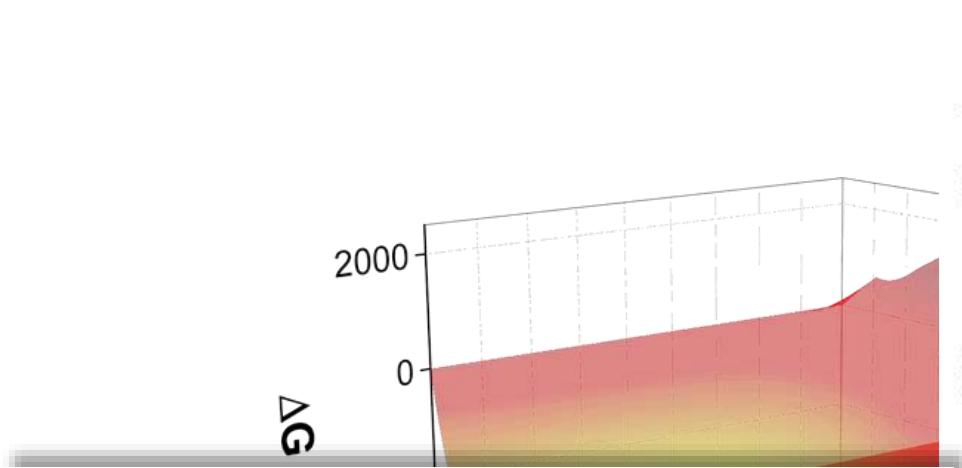
1. GB is NOT part of equilibrium, it is frozen in
2. BUT local composition / segregation is in equilibrium (Gibbs isotherm)
3. YET, segregants can interact (Fowler & Guggenheim, Hart, Guttmann isotherms)

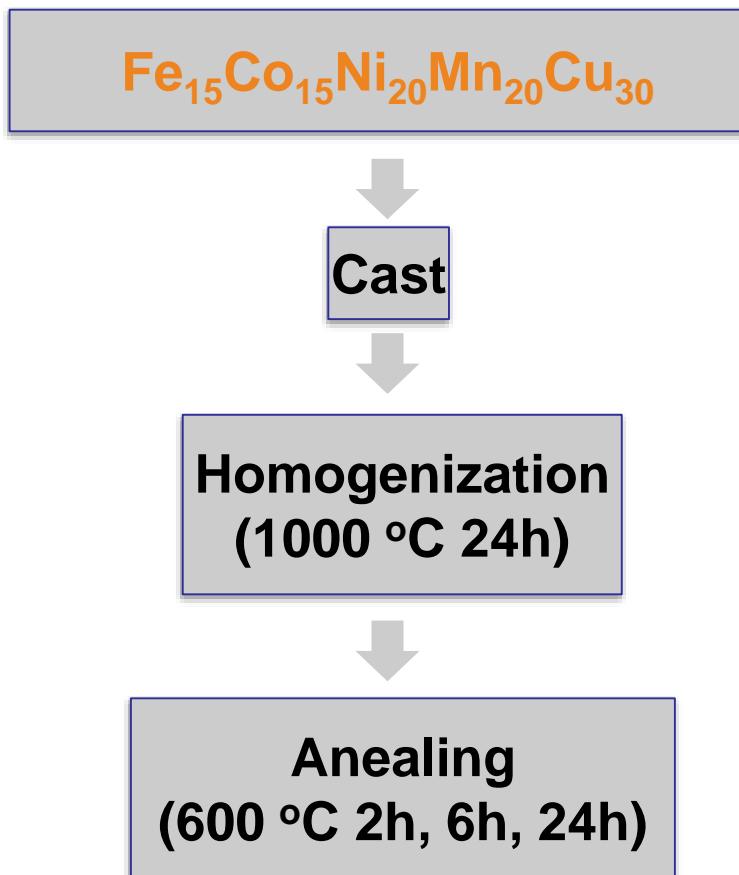
'Heterogeneous' nucleation:

- Different compositional equilibrium working point
- Different driving force
- Different interface energy (isotherm)
- GP-like precursor states or other...



Defect decoration & thermodynamics

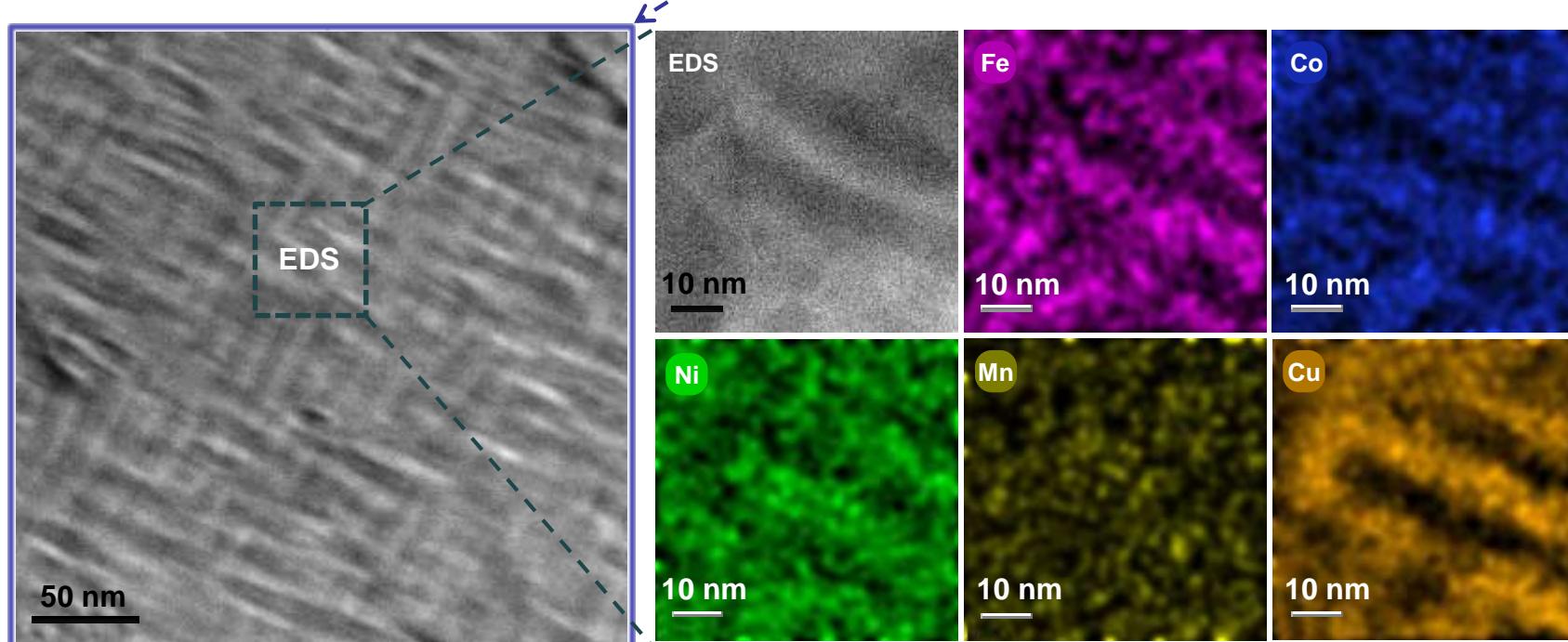
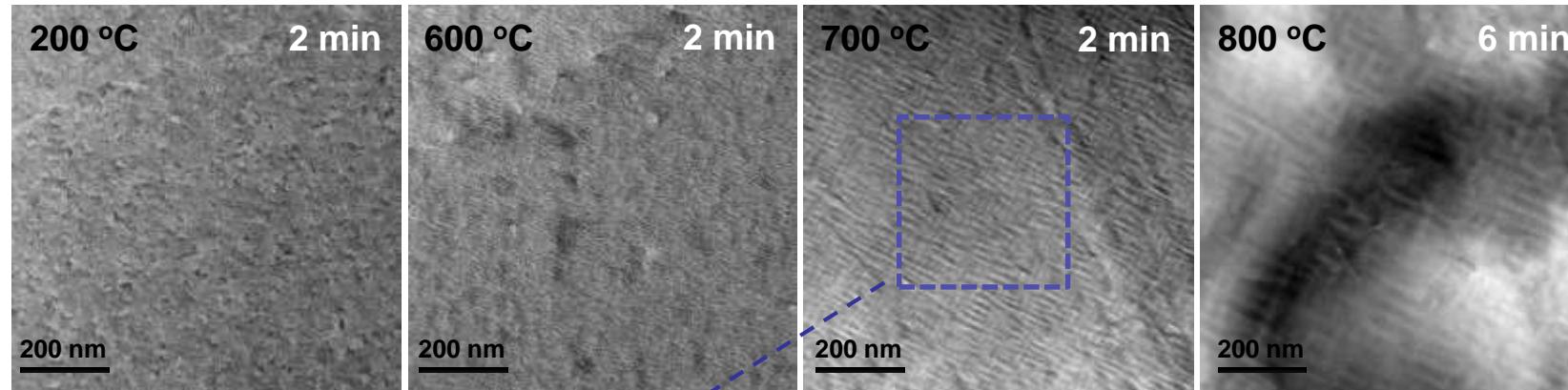




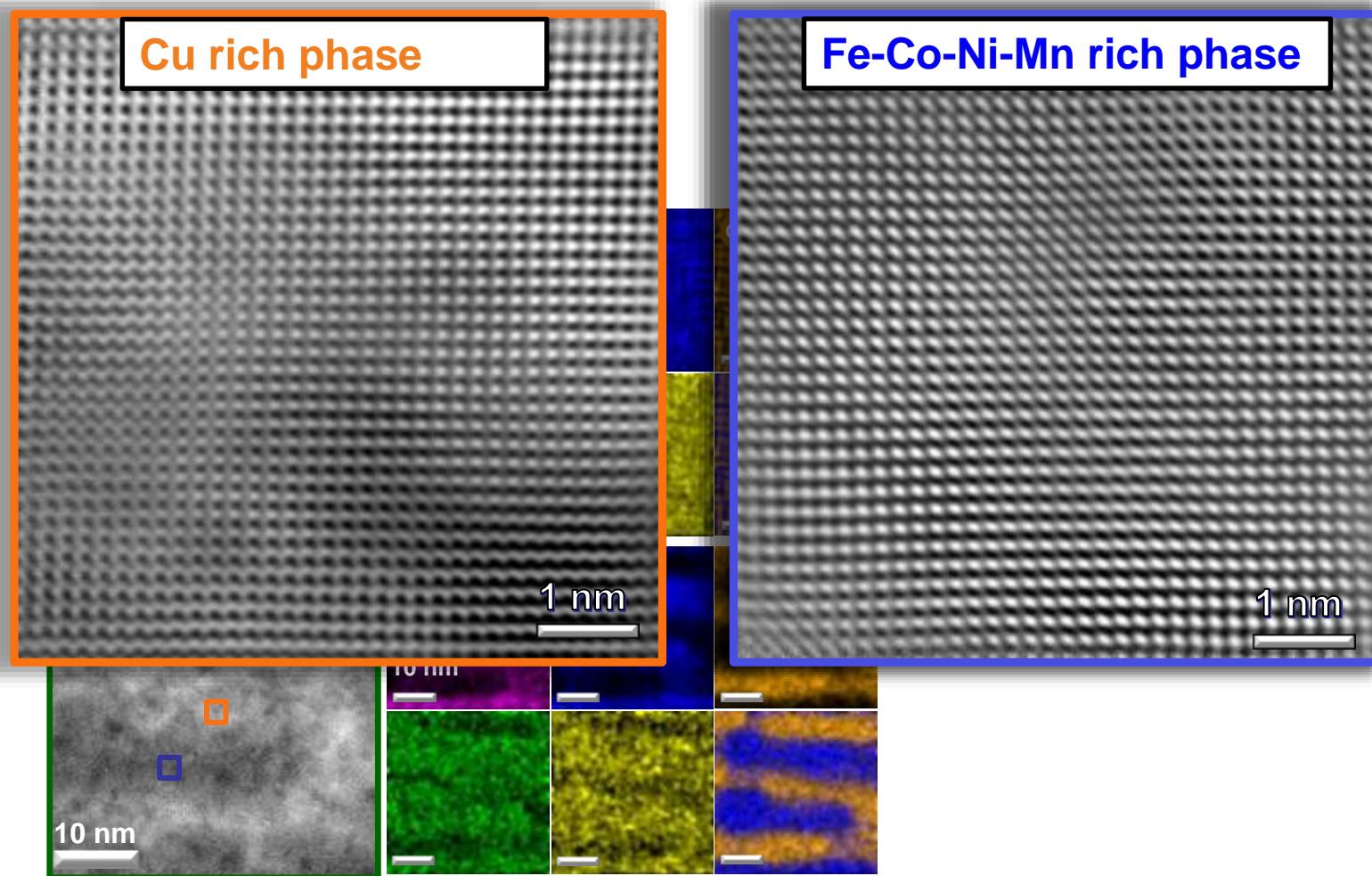
Bulk spinodal: tuning for ferromagnetism



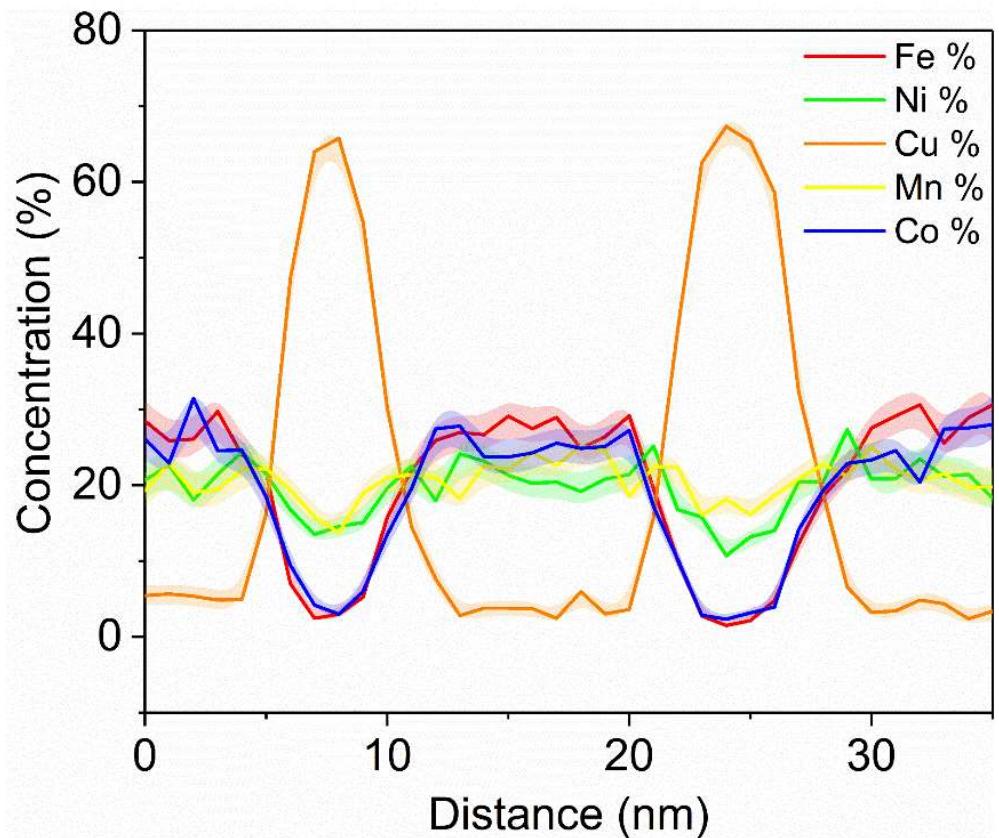
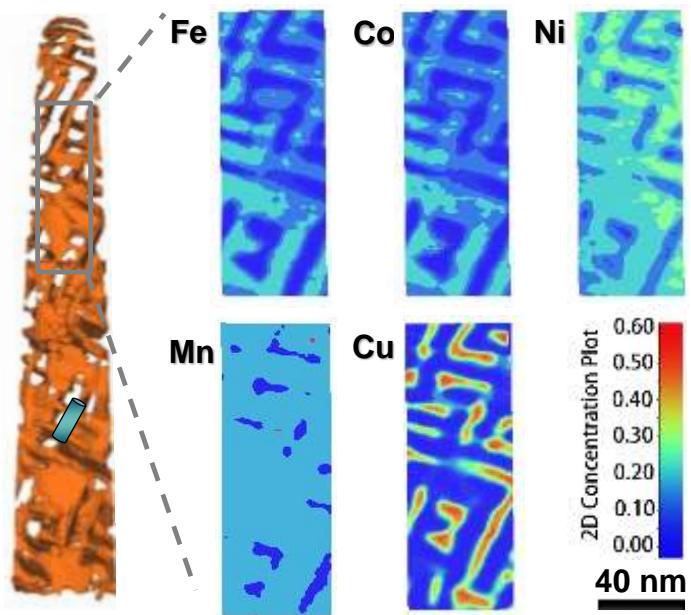
Temperature, in-situ heating



FeCoNiMnCu annealed at 600°C for 6 h



FeCoNiMnCu annealed at 600°C for 6 h



FeCoNiMnCu annealed at 600°C



Annealing time →

0 h



2 h



6 h



24 h



240 h



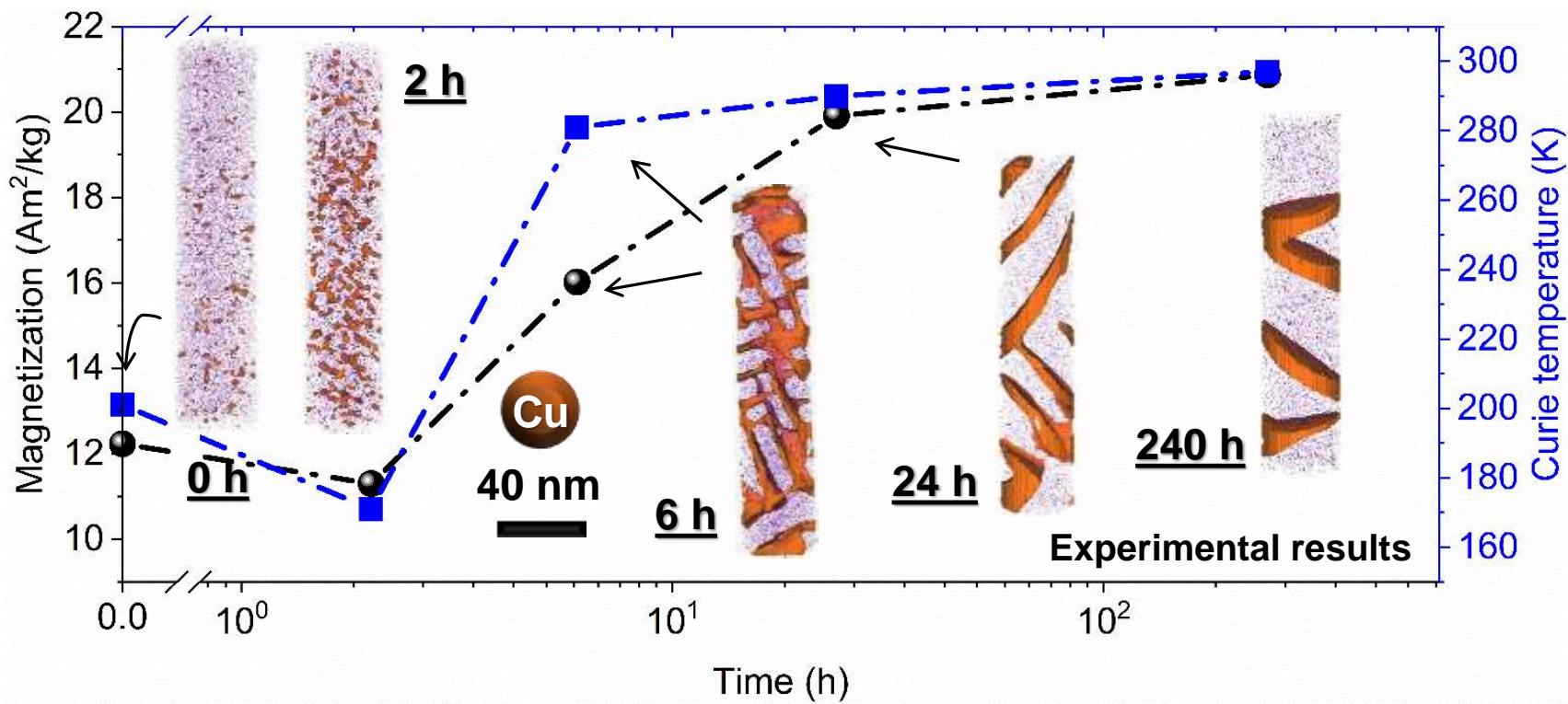
Iso-surface:
50 at% Cu



20 nm



FeCoNiMnCu annealed at 600°C



- 
- Metastability alloy design (displacive, spinodal)
 - Multicomponent thermodynamics, also at confined scales

- 
- Combine mechanical properties with magnetism, corrosion, H-resistance, ...

Thank you for the attention

Kindly sponsored by:
European Research Council
Deutsche Forschungsgemeinschaft
Alexander von Humboldt-Stiftung
Bundesministerium Bildung und Forschung